

# Geometric Study of Polymer Embedded Micro Thermoelectric Cooler with Optimized Contact Resistance

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Micro-thermoelectric devices ( $\mu$ TEDs) are used for bio-medical applications, powering internet-of-things devices, and thermal management. For such applications,  $\mu$ TEDs need to have a robust packaging so that the devices can be brought in direct thermal contact with the target heat sink and source. The packaging technology developed for macroscopic modules needs improvement as it cannot be applied to  $\mu$ TEDs due to a large thermal resistance between the capping material and the device which deteriorates its performance. In this work,  $\mu$ TEDs with high net cooling temperature are fabricated by optimizing the contact resistance and device design combined with a novel packaging technique that is fully compatible with on-chip integration. The simulations and experiments demonstrate that the additional thermal loss caused by the packaging leads to an only marginal decrease in the net cooling temperature. The devices achieve a high net cooling temperature of 10.8 K without packaging and 9.6 K with packaging at room temperature. The packaging only slightly increases the thermal response time of the devices, which also shows an extremely high reliability of over 85 million cooling cycles. This simple packaging technique together with robust device performance is a step toward wide-spread application of  $\mu$ TEDs.

any moving parts and are thus maintenance free, and also environment-friendly.<sup>[1–4]</sup> They are used for the thermal management of parts of electric circuits<sup>[5]</sup> or optoelectronic devices,<sup>[6]</sup> cooling of power electronics,<sup>[7]</sup> powering devices for the internet-of-things,<sup>[8]</sup> and body-powered wearable electronics.<sup>[9–11]</sup> With increased reliability and accurate temperature control, they could also be effective replacements for macroscopic thermoelectric devices (TEDs) in medical applications,<sup>[12–14]</sup> for example DNA replication<sup>[15]</sup> or heating and cooling experiments for treating low-grade tissue injuries.<sup>[16]</sup>

Early  $\mu$ TEDs were fabricated by J.-P. Fleurial and co-workers.<sup>[17]</sup> During the past two decades, as the approaches to fabrication have evolved continuously, so has the performance of  $\mu$ TEDs.<sup>[1,2,18–20]</sup> In 2003, Snyder et al. fabricated a cross plane  $\mu$ TED using electrochemical deposition, which showed a maximum cooling of 2 K at

110 mA.<sup>[2]</sup>  $\mu$ TEDs based on superlattice materials were later fabricated but were not characterized as coolers.<sup>[21]</sup> In 2007, Huang et al. fabricated a device using MEMS technology in combination with electroplating and reported a maximum cooling of

## 1. Introduction

Micro-thermoelectric devices ( $\mu$ TEDs) directly convert heat fluxes into electrical energy, or vice versa. They work without

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0.7 K using infrared imaging.<sup>[22]</sup> Later in 2008, the same group improved the net cooling temperature to 1.2 K.<sup>[23]</sup> In 2013, Roth et al. developed a flip chip bonding process and fabricated a device using two different substrates but was not characterized as coolers.<sup>[24]</sup> A startup company called Micropelt fabricated  $\mu$ TEDs using flip chip bonding in combination with sputtering, which proves to be an expensive technique.<sup>[25]</sup> Later in 2021, a thin film  $\mu$ TED with extreme heat flux handling was demonstrated.<sup>[26]</sup> In this work, we report on the fabrication of  $\mu$ TEDs based on electrochemical deposition, a cost-effective method that works at room temperature without vacuum to synthesize materials with desired composition and crystalline structure. The most important performance measure of a thermoelectric material is its thermoelectric figure of merit  $zT = (S^2 \sigma \kappa^{-1}) T$ , where  $S$  is the Seebeck coefficient;  $\sigma$  is the electrical conductivity;  $\kappa$  is the total thermal conductivity; and  $T$  is the absolute temperature.<sup>[27,28]</sup> The performance of thermoelectric materials has improved dramatically in recent years: for instance,  $zT$  was improved from around 1<sup>[29]</sup> in bismuth telluride and bismuth selenide systems to 2.6 in SnSe single crystals.<sup>[30]</sup> Although the performance of thermoelectric materials has improved greatly, that has not necessarily led to a substantial improvement in the performance of the devices themselves. This is because the performance also depends on the design and assembly of the TEDs.<sup>[31]</sup> The performance of TEDs themselves is characterized by the device figure of merit  $ZT = (S_p - S_n)^2 T (RK)^{-1}$ , where  $S_p$  and  $S_n$  are the Seebeck coefficients of p-type and n-type legs;  $R$  is the electrical resistance of the device (including the contact resistance and the resistance of the metallization); and  $K$  is the thermal conductance of the device (including the thermal conductance of contact material and that of the metallization).<sup>[32]</sup> For optimized thermoelectric performance, aside from highest possible  $zT$ , it is also important to have excellent electrical contacts, thermal interfaces, mechanical properties, interconnects, and packaging technology.

The importance of packaging technology is often neglected in the fabrication of thermoelectric coolers (TECs) generally and micro-TECs ( $\mu$ TECs) in particular. This insulating fillers in the gaps between the thermoelectric legs mechanically support the devices. The filling material prevents oxidation, improving the chemical stability. Because the packaging technology of macroscopic modules entails a large thermal resistance between the capping material and the device, it cannot be applied to  $\mu$ TECs without deteriorating their performance substantially.

Previously, our group fabricated a  $\mu$ TEC by a combination of electrochemical deposition and photolithography. This work assessed transient response and cooling stability as well as cycling reliability.<sup>[1]</sup> Here we show that by effectively optimizing the geometry and the contact resistance, we were able to obtain a greatly improved net cooling temperature by 80% as compared with previous work. Further, the fabricated device showed a record high cycling stability of 100 million cycles. We were able to reduce the device's contact resistance such that we obtained a cooling power density close to theoretical maximum when the contact resistance would be zero. As of our knowledge, this is the highest reported net cooling temperature and cycling stability of a  $\mu$ TEC fabricated using electrochemical deposition. We then embedded the  $\mu$ TECs in a low- $\kappa$  polymer filling material. We found that the polymers gave mechanical integrity and chemical stability to the devices with an only marginal decrease in the net cooling temperature. Polymer

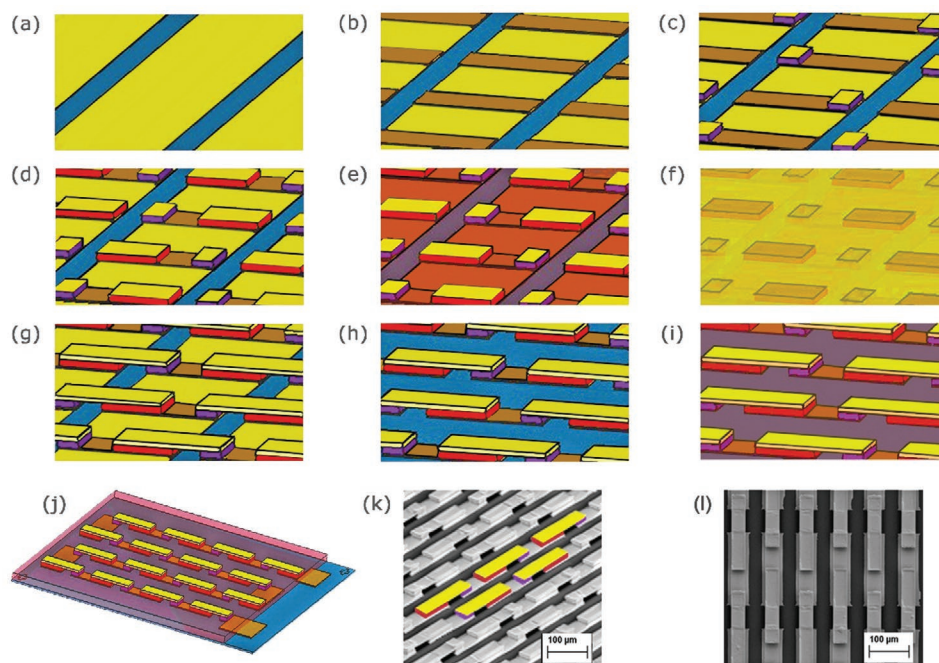
embedded micro devices pave the way for a widespread application of thermoelectrics in modern electronics and for potential applications in bio medical field, that is, to study temperature effects even on single biological cells.

## 2. Experimental Section

### 2.1. Fabrication Technology

The devices were fabricated by using multiple steps of photolithography and electrochemical deposition. All photolithographic steps were done using a laserwriter ( $\mu$ PG 101, Heidelberg Instruments GmbH, Germany, 375 nm irradiation wavelength). The device was fabricated on a Si substrate, which was coated by a  $\text{Si}_3\text{N}_4$  layer with a thickness of 100 nm. 5 nm Cr and 100 nm Au were sputtered on the substrate as the adhesion layer and as seed layer, respectively, for the electrochemical deposition. For the first lithography, an adhesion promoter (TI Prime, MicroChemicals GmbH, Germany), photoresist (AZ5214E, MicroChemicals GmbH, Germany), and developer (AZ 726 MIF, MicroChemicals GmbH, Germany) were used to pattern alignment markers and to structure the metallization layer of Cr/Au on desired parts of the substrate for the subsequent etching (Figure 1a). The Au was etched using a KI/I<sub>2</sub> solution and Cr was etched with a commercial Cr etchant solution (TechniEtch Cr01, MicroChemicals GmbH, Germany). After etching, the photoresist was removed using n-methyl pyrrolidone (NMP) and the remaining photoresist residues were removed by O<sub>2</sub> plasma. For the next photolithography (Figure 1b), a 10  $\mu\text{m}$  thick photoresist (AZ9260, MicroChemicals GmbH, Germany) and developer (AZ 400K [1:4 solution], MicroChemicals GmbH, Germany) were used. After patterning and developing, 3  $\mu\text{m}$  thick Au was electrochemically deposited on the patterned structures.

After removing the photoresist residues, the next structuring was done with 24  $\mu\text{m}$  thick photoresist. Here, AZ9260 was used twice to get the desired thickness. n-type  $\text{Bi}_2(\text{Te}_{0.95}\text{Se}_{0.05})_3$  (in the following referred to as BiTeSe) was electrochemically deposited into the cavities and the composition was determined by EDX. The thickness of the deposited material was 10  $\mu\text{m}$ . After the n-type deposition, the sample was electrochemically deposited with 1  $\mu\text{m}$  of Au (Figure 1c). Subsequently, p-type Te was deposited, together with 1  $\mu\text{m}$  of electrodeposited Au on top of the Te legs (Figure 1d). After the deposition of thermoelectric legs, multiple photolithographic steps were used for depositing the top Au bridge. 24  $\mu\text{m}$  thick AZ9260 was spin-coated on the substrate. After spin-coating, the photoresist was leveled at the same height as that of the thermoelectric legs by exposure to a smaller dose using  $\mu$ PG 101 and by controlling the development time (Figure 1e). After leveling, 40 nm Au film was sputtered on the sample, acting as a seed layer for the subsequent electrochemical deposition process and preventing the removal of leveled photoresist during further steps (Figure 1f). A 10- $\mu\text{m}$ -thick photoresist was spin-coated on top of the sputtered Au film. Photolithography was performed for structuring the top bridge (Figure 1g). 6- $\mu\text{m}$ -thick Au was electrochemically deposited into the cavities (see subchapter 4.2), followed by lift-off. After lift-off, a photolithographic process was done to structure the bottom contact on the substrate (Figure 1h).



**Figure 1.** Schematic fabrication process of free-standing and embedded  $\mu$ TECs. a) Structuring the Au seed layer on a Si/Si<sub>3</sub>N<sub>4</sub> substrate; b) Electrochemical deposition of 3  $\mu$ m thick bottom Au electrode; c) Electrochemical deposition of 10  $\mu$ m thick n-type BiTeSe and subsequent plating 1  $\mu$ m thick Au top electrode on BiTeSe; d) Electrochemical deposition of 10  $\mu$ m thick p-type Te and subsequent plating 1  $\mu$ m thick Au top electrode on Te; e) Levelling the thermoelectric legs with AZ9260; f) Sputtering thin layer of Au; g) electrochemical deposition of 6  $\mu$ m thick top Au; h) Free-standing  $\mu$ TEC; i) Polymer embedded  $\mu$ TEC; j) Graphical representation of polymer embedded  $\mu$ TEC; k) SEM image of  $\mu$ TEC (side view); l) SEM image of  $\mu$ TEC (top view).

To embed the as-fabricated free-standing device, AZ9260 was spin-coated twice. After spin-coating, the top Au portion was exposed and developed. Then leveling was done by exposure with a partial dose and by controlling the development time, thus achieving the polymer height in the desired range of the height of the device (Figure 1j). Finally, the whole device was embedded in a polymer matrix, that is, the polymer matrix forms a uniform top surface with the same height as the thermoelectric legs.

## 2.2. Electrochemical Deposition of Au, Bi<sub>2</sub>(Te<sub>0.95</sub>Se<sub>0.05</sub>)<sub>3</sub> and Te

Prior to all electrochemical depositions, the device was cleaned in O<sub>2</sub> plasma using a reactive ion etching (RIE) device (SI220 from Sentech Instruments GmbH, Germany) with an O<sub>2</sub> gas flow of 25 sccm at a power of 50 W for 2 min.

Thick Au films were used as bottom and top contacts for the device. Electrochemical deposition was used to get thicker Au films. The commercial electrolyte NB Semiplat Au 100 (NB Technologies GmbH, Germany) was used for the electrochemical deposition of thick Au films. The electrochemical deposition of Au was carried out in galvanostatic mode in a two-electrode configuration with a Pt counter electrode at room temperature using a VSP modular five channels potentiostat (Bio-Logic SAS, France). The current density was fixed at 18.35  $\mu$ Acm<sup>-2</sup>.

BiTeSe and Te were used as n-type and p-type thermoelectric materials. Both these materials were electrochemically synthesized from HNO<sub>3</sub> (Merck, 69%) based electrolytes. The chemicals used were Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O (Merck, 99.999%), TeO<sub>2</sub> (Merck, 99.5%), SeO<sub>2</sub> (Strem chemicals, 99.8%). The electrolyte

for BiTeSe was prepared by dissolution of 10 mm TeO<sub>2</sub> in 1M HNO<sub>3</sub> followed by the addition of 10 mm Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O and 1.1 mm SeO<sub>2</sub>. For the preparation of p-type Te electrolyte, 10 mm of TeO<sub>2</sub> was dissolved in 1M HNO<sub>3</sub>. Just before electrochemically depositing Te, sodium dodecyl sulfate (SDS) was added as surfactant to the electrolyte to make the deposited thick film smoother.

The n-type and p-type electrochemical depositions were performed under pulsed potentiostatic conditions in a three-electrode configuration with a Pt counter electrode and Ag/AgCl reference electrode at room temperature using a modular five channels potentiostat. For the pulsed electrochemical deposition of BiTeSe, the deposition potential of  $E_{\text{on}} = -0.015$  V was applied for  $t_{\text{on}} = 10$  ms, followed by a reverse potential  $E_{\text{off}} = 0.186$  V for  $t_{\text{off}} = 50$  ms. For the pulsed electrochemical deposition of Te, the deposition potential of  $E_{\text{on}} = -0.075$  V was applied for  $t_{\text{on}} = 10$  ms, followed by a reverse potential  $E_{\text{off}} = 0.180$  V for  $t_{\text{off}} = 50$  ms.

To improve the quality of the thermoelectric materials, the oxygen saturation of the electrolyte was modified by bubbling the electrolyte with N<sub>2</sub> gas (5N quality) for 20 min prior to deposition for  $\mu$ TEC<sub>4</sub>.

## 2.3. Thermoreflectance Microscopy

The characterization of the devices was done with charge coupled device based thermoreflectance microscopy. Thermoreflectance microscopy was a thermal imaging technique that provided non-contact, non-invasive measurements with high spatial and thermal resolution. This technique was used to

visualize the surface-temperature distribution of active devices by detecting the relative variations in the reflectivity of a sample in response to variations in the temperature.<sup>[33]</sup>

$$\Delta T = \frac{\Delta I_R}{I_{R0} \cdot C_{th}} \quad (1)$$

where  $\Delta I_R$  and  $I_{R0}$  are the change and absolute value in the thermal reflectivity and  $C_{th}$  is thermorefectance coefficient. For every material,  $C_{th}$  needs to be calibrated, separately.<sup>[33]</sup>

The thermorefectance microscope NT100 from Microsanj LLC, USA, with transient resolution of 10  $\mu$ s and temperature resolution of 1 K, was used in this study. The calibration was done using the electrochemically deposited Au, resulting in a  $C_{th}$  value of  $-2.8 \times 10^{-4} \text{ K}^{-1}$ . Since the reflectivity strongly depends on the surface roughness, all the values were taken from the top Au bridge. To characterize the dependence of electric current on the cooling, a pulsed current was passed and the change in temperature was averaged for at least 100 s. For the thermal transient measurements, a pulsed electric current was applied and the net cooling temperature was observed at different time delay. For the cycling reliability measurements, a pulsed electric current was applied, with a pulse time of 5 ms (1 ms on and 4 ms off) and an amplitude of the electric current which corresponds to the maximum net cooling temperature of the devices. The error of the temperature measurement using thermorefectance technique was  $\approx 4\%$ .

## 2.4. Finite Element Simulation

FEM simulations were carried out with the commercially available COMSOL Multiphysics software package. In this manner it was possible to calculate approximations of the coupled heat and electric flux equations involved in thermoelectricity.

## 3. Results and Discussion

### 3.1. Device Optimization

The performance of a TED is determined primarily by  $ZT$ , which can be improved for a given material by optimizing the

contact resistance and device geometry.<sup>[34]</sup> Whereas  $S_p$  and  $S_n$  are material properties that are independent of the geometry,  $R$  and  $K$  depend on the geometry of the device: their product should be minimum for a maximum value of  $ZT$ .

$R = \rho_p A_p^{-1} L_p + \rho_n A_n^{-1} L_n$ , where  $L$  is the height,  $A$  is the cross-sectional area, and  $\rho$  is the electrical resistivity.  $K = \kappa_p A_p L_p^{-1} + \kappa_n A_n L_n^{-1}$ , where  $\kappa$  is the thermal conductivity, and the indices  $n$  and  $p$  correspond to  $n$ -type and  $p$ -type legs, respectively. To minimize the product  $RK$ , the following condition has to be met:

$$\frac{L_n \cdot A_p}{L_p \cdot A_n} = \sqrt{\frac{\kappa_n \cdot \rho_p}{\kappa_p \cdot \rho_n}} \quad (2)$$

Both legs of the TED should have the same length to ensure a good thermal contact between the thermoelectric elements and the heat source and sink. As a result, the cross-sectional area is the only geometric parameter that can be tuned for given materials properties.

In this work, we use  $n$ -type  $\text{Bi}_2(\text{Te}_{0.95}\text{Se}_{0.05})_3$  and  $p$ -type  $\text{Te}$  as thermoelectric materials (see Table 1 for their properties).

To understand how the device geometry influences its cooling performance and thereby changes  $ZT$ , we fabricated two  $\mu$ TECs with 160 leg pairs each and compared their performance. The details of the geometry of each  $\mu$ TEC are shown in Table 1. Whereas the  $A_p A_n^{-1}$  ratio of  $\mu$ TEC\_A was with 3.58 far from its optimal ratio of 1.37, the  $A_p A_n^{-1}$  ratio of  $\mu$ TEC\_B was found to be 2.64, which is close to its ideal ratio of 1.84. The deviation from the optimal ratio results from inaccuracies in the lithographic processing. In addition, the electrolyte used for the fabrication of  $\mu$ TEC\_B was bubbled with  $\text{N}_2$  before the electrochemical deposition.

The in-plane electrical conductivity of the materials was characterized using the van der Pauw method (Section I and Figures S1,S2, Supporting Information). Considering an anisotropic transport in  $\text{BiTeSe}$  of 2.3<sup>[37]</sup> between the in-plane and cross-plane direction we obtained an electrical conductivity of 700  $\text{S cm}^{-1}$  and 1090  $\text{S cm}^{-1}$  for the  $\text{BiTeSe}$  without  $\text{N}_2$  bubbling and with  $\text{N}_2$  bubbling, respectively. The electrical conductivity of 700  $\text{S cm}^{-1}$  of the  $\text{BiTeSe}$  sample without  $\text{N}_2$  bubbling, is in good agreement with the electrical conductivity reported by Schumacher et al<sup>[35]</sup> and is well within the error range of

**Table 1.** Materials properties and device design.

Materials	Section	Device	Dimension [width $\times$ length $\times$ height] [ $\mu\text{m} \times \mu\text{m} \times \mu\text{m}$ ]	$\sigma$ [ $\text{S cm}^{-1}$ ]	$S$ [ $\mu\text{V K}^{-1}$ ]	$\kappa$ [ $\text{W m}^{-1} \text{K}^{-1}$ ]
BiTeSe	n-type	$\mu$ TEC_A	$33 \times 33 \times 7.5$	700	-60	1.2 <sup>[35]</sup>
		$\mu$ TEC_B	$33 \times 33 \times 9$	1090	-50	1.4
Te	p-type	$\mu$ TEC_A	$33 \times 118 \times 7.5$	250	210	1.8 <sup>[36]</sup>
		$\mu$ TEC_B	$33 \times 87 \times 9$	250		1.8 <sup>[36]</sup>
Au	Bottom contact	$\mu$ TEC_A	$33 \times 33/118 \times 3$	406 000	6.5 <sup>a)</sup>	317 <sup>a)</sup>
		$\mu$ TEC_B	$33 \times 33/87 \times 3$			
Au	Top contact	$\mu$ TEC_A	$33 \times 201 \times 6$	406 000	6.5 <sup>a)</sup>	317 <sup>a)</sup>
		$\mu$ TEC_B	$33 \times 170 \times 6$			

<sup>a)</sup>Value from database of COMSOL multiphysics.



the electrical conductivity reported in other works for BiTeSe samples.<sup>[38]</sup> With the incorporation of N<sub>2</sub> bubbling an increase in the electrical conductivity was observed. To confirm the cross-plane electrical conductivity and the assumed anisotropic ratio, the BiTeSe sample with N<sub>2</sub> bubbling was additionally characterized using the Cox-Strack method (Section X, Supporting Information). With the Cox-Strack method, a cross-plane electrical conductivity of (1145 ± 246) S cm<sup>-1</sup> was obtained, which is in good agreement with the electrical conductivity of 1090 S cm<sup>-1</sup> obtained from the van der Pauw method. The thermal conductivity of the BiTeSe without N<sub>2</sub> bubbling was taken as 1.2 W m<sup>-1</sup> K<sup>-1</sup>.<sup>[35]</sup> Considering the Wiedemann Franz law the thermal conductivity of the BiTeSe with N<sub>2</sub> bubbling was slightly increased to 1.4 W m<sup>-1</sup> K<sup>-1</sup> (Section II, Supporting Information). The Seebeck coefficient was measured using a Potential Seebeck Microprobe system. For the BiTeSe without N<sub>2</sub> bubbling, the Seebeck coefficient was found to be -60 μV K<sup>-1</sup>. While for the BiTeSe with N<sub>2</sub> bubbling a slight decrease in the Seebeck coefficient to -50 μV K<sup>-1</sup> was observed.

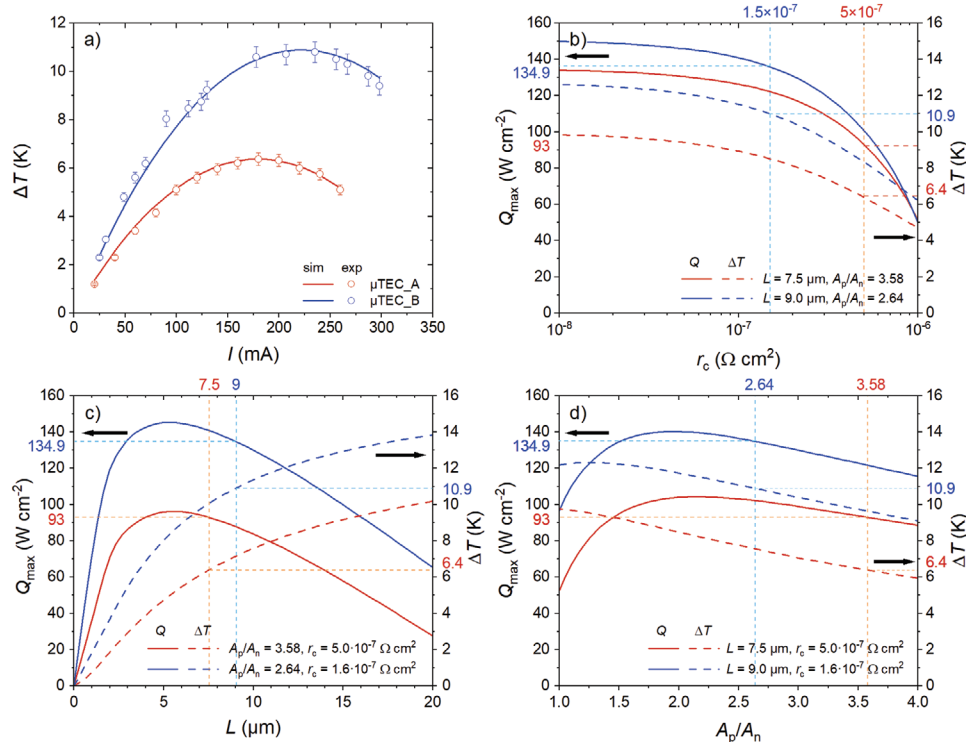
The electrical transport properties of the electrochemically deposited material were improved by purifying the electrolytes prior to the electrochemical deposition. Oxygen, once incorporated into the TE material, is known to deteriorate the materials power factor  $PF = S^2 \sigma$ .<sup>[39]</sup> Since the physically dissolved O<sub>2</sub> can be easily integrated into the electrochemically deposited material,<sup>[40]</sup> its replacement by N<sub>2</sub> helps to improve the electrical transport properties.<sup>[41]</sup> The electrolytes were hence bubbled with N<sub>2</sub> for 20 min so that the physically dissolved O<sub>2</sub> gas within the electrolytes would be replaced by N<sub>2</sub> gas. The EDX data of

the electrochemically deposited BiTeSe showed reduction in the atomic percent of oxygen from 6.9 at% to 2.8 at% after N<sub>2</sub> bubbling (Section III, Figure S3, and Table S1, Supporting Information). The BiTeSe with N<sub>2</sub> bubbling showed a higher electrical conductivity and thermal conductivity in combination with a slightly decreased absolute Seebeck coefficient. Thus, the reduction in oxygen improved the electrical transport properties of the BiTeSe but resulted in a slightly decreased material  $zT$  from 0.063 to 0.058.

The obtained electrical conductivity and Seebeck coefficient of the electrochemically deposited Te were found to be 250 S cm<sup>-1</sup> and 210 μV K<sup>-1</sup>, respectively. The thermal conductivity was taken as 1.8 W m<sup>-1</sup> K<sup>-1</sup>.<sup>[36]</sup> The bubbled Te sample was also studied but no impact on the thermoelectric properties was observed.

Figure 2a) shows that the variation in the geometries of the μTECs led to a shift in the value of the electric current. The maximum cooling for μTEC\_A and μTEC\_B was seen at 180 and 235 mA, respectively. When the electric current was increased further, the cooling decreased because of the effect of joule heating in the device. As compared with μTEC\_A, the maximum net cooling temperature of μTEC\_B showed an increase from 6.4 to 10.8 K. The thermal image of both devices at maximum net cooling temperature can be found in the supporting information (Section IV and Figure S4, Supporting Information). The device  $ZT$  of μTEC\_A and μTEC\_B obtained by  $ZT = 2 \Delta T T^{-1}$  equals 0.044 and 0.074, respectively.

With the combined effect of geometric optimization and N<sub>2</sub> bubbling, a 69% improvement in the net cooling temperature



**Figure 2.** Cooling Performance study of freestanding devices. a) The net cooling temperature of μTEC\_A and μTEC\_B. Study of variation in the net cooling temperature and cooling power density of device with b) fixed height and  $A_p/A_n$  ratio by varying the  $r_c$  of the device c) fixed  $r_c$  and  $A_p/A_n$  ratio by varying the height of the device d) fixed  $r_c$  and height by varying the  $A_p/A_n$  ratio of the device.

and 68% improvement in the device  $ZT$  was observed, although we observed a slight decrease in the materials  $zT$ . We mainly attribute this improvement to the device's reduced contact resistance ( $r_c$ ) and optimized geometry. For the simulations shown in Figure 2a) the material properties from Table 1 were used. To match the simulation to the experimental results the  $r_c$  of the  $\mu\text{TEC}_A$  had to be adjusted to  $5 \times 10^{-7} \Omega \text{ cm}^2$  for  $\mu\text{TEC}_A$  and  $1.6 \times 10^{-7} \Omega \text{ cm}^2$  for  $\mu\text{TEC}_B$ . This is in good agreement with the experimentally found  $r_c$  of  $(1.6 \pm 1.4) \cdot 10^{-7} \Omega \text{ cm}^2$  of the  $\text{N}_2$  bubbled sample, which was characterized with the Cox–Strack method (Section X, Supporting Information), and is well within recently published results of Corbett et al.<sup>[42]</sup> We attribute the reduction of the  $r_c$  to the purification of the electrolytes prior to the electrochemical deposition, that is, for the  $\text{N}_2$  bubbled  $\mu\text{TEC}_B$ , the oxygen content was reduced, which might lead to the improved contact between the thermoelectric film and the top electrode. In addition to the increase in the net cooling temperature, a broader range of optimum current, which is directly related to the increased device figure of merit, was obtained making the device more suitable for applications.

In Figure 2b the net cooling temperature and the maximum cooling power density  $Q_{\text{max}}$  at optimal electrical current as function of  $r_c$  simulated for an  $A_p A_n^{-1}$  ratio of 3.58 and a leg height of  $7.5 \mu\text{m}$  corresponding to  $\mu\text{TEC}_A$  and an  $A_p A_n^{-1}$  ratio of 2.64 and a leg height of  $9 \mu\text{m}$  corresponding to  $\mu\text{TEC}_B$  is presented. Details of the simulation can be found in the supporting information (Section V and Figure S5, Supporting Information). With increasing  $r_c$  a decrease of  $Q_{\text{max}}$  and the net cooling temperature of the device was observed. The effective device figure of merit  $Z_e$  including  $r_c$  is given by  $Z_e = Z(1 + r_c \rho^{-1} L^{-1})^{-1}$ . Therefore,  $Z_e$  decreases as  $r_c$  increases, resulting in a decreased net cooling temperature and  $Q_{\text{max}}$ . The  $r_c$  of  $\mu\text{TEC}_A$  and  $\mu\text{TEC}_B$  are marked in Figure 2b) with blue and red dashed lines, respectively.  $Q_{\text{max}}$  was increased by 45% from  $93 \text{ W cm}^{-2}$  for  $\mu\text{TEC}_A$  to  $134.9 \text{ W cm}^{-2}$  for  $\mu\text{TEC}_B$ . Note that for the simulations in Figure 2  $Q_{\text{max}}$  was defined as maximum power output per leg pair at optimal electrical current divided by the area of the top contact of one leg pair. Due to the decreased  $r_c$  from  $5 \times 10^{-7} \Omega \text{ cm}^2$  to  $1.6 \times 10^{-7} \Omega \text{ cm}^2$ , the maximum net cooling temperature of  $\mu\text{TEC}_B$  was increased by 31% from  $8.3$  to  $10.9 \text{ K}$  and  $Q_{\text{max}}$  was increased from  $100.6 \text{ W cm}^{-2}$  to  $134.9 \text{ W cm}^{-2}$ . With the same  $r_c$  as  $\mu\text{TEC}_B$ , the net cooling temperature of  $\mu\text{TEC}_A$  would be increased by 33% to  $8.5 \text{ K}$  and  $Q_{\text{max}}$  would increase to  $122.3 \text{ W cm}^{-2}$ .

In Figure 2c the simulated net cooling temperature and the  $Q_{\text{max}}$  in dependency of the height of the thermoelectric legs with a constant  $A_p A_n^{-1}$  ratio of 3.6 and  $r_c$  of  $5 \times 10^{-7} \Omega \text{ cm}^2$  corresponding to  $\mu\text{TEC}_A$  and a constant  $A_p A_n^{-1}$  ratio of 2.6 and  $r_c$  of  $1.6 \times 10^{-7} \Omega \text{ cm}^2$  corresponding to  $\mu\text{TEC}_B$  are shown. As the height of the thermoelectric legs increased, the thermal conductivity of the device decreased, which leads to an increased net cooling temperature. While  $Q_{\text{max}}$  initially increased until it reached a maximum at a thermoelectric leg height of  $\approx 5 \mu\text{m}$  for both  $\mu\text{TECs}$ . With further increased leg heights  $Q_{\text{max}}$  started decreasing again. The trend results from the increased electrical resistance with increasing leg height, which leads to a reduced optimal current. As a result, Joule heating reached a maximum at an intermediate height.  $Q_{\text{max}}$  is determined by the balance of Joule heating, Peltier cooling, which decreases with the current,

and the thermal conductance, which follows a  $1/L$  dependency. Therefore, choosing the optimal thickness of the thermoelectric legs is a trade-off between maximum net cooling temperature and  $Q_{\text{max}}$ . The heights of  $7.5$  and  $9 \mu\text{m}$  of  $\mu\text{TEC}_A$  and  $\mu\text{TEC}_B$  are marked in Figure 2c) with blue and red dashed lines, respectively. By increasing the leg height of  $\mu\text{TEC}_B$  from  $7.5$  to  $9 \mu\text{m}$ , the maximum net cooling temperature was increased by 9% from  $10$  to  $10.9 \text{ K}$  but in the same time  $Q_{\text{max}}$  decreased by 6%. With the same leg height as  $\mu\text{TEC}_B$ , the net cooling temperature of  $\mu\text{TEC}_A$  would be increased by 19% to  $7.6 \text{ K}$  with a slightly reduced  $Q_{\text{max}}$  of  $87.8 \text{ W cm}^{-2}$ .

Figure 2d) shows the simulated net cooling temperature and  $Q_{\text{max}}$  depending on the  $A_p A_n^{-1}$  ratio with a constant height of  $7.5 \mu\text{m}$  and  $r_c$  of  $5 \times 10^{-7} \Omega \text{ cm}^2$  corresponding to  $\mu\text{TEC}_A$  and a constant height of  $9 \mu\text{m}$  and  $r_c$  of  $1.6 \times 10^{-7} \Omega \text{ cm}^2$  corresponding to  $\mu\text{TEC}_B$ . When the  $A_p A_n^{-1}$  ratio is increased, the net cooling temperature first increased until it reaches a maximum when the thermal and electrical resistance are balanced, and then decreases again. Since the simulation also includes the contribution from the contact resistance and the resistance of the contacts, this result deviates from the analytical obtained result from Equation (1). The additional resistance shifts the optimal  $A_p A_n^{-1}$  ratio for  $\mu\text{TEC}_A$  from 1.37 to 1 and for  $\mu\text{TEC}_B$  from 1.84 to 1.2. The  $Q_{\text{max}}$  also increased until it reached a maximum, but at a higher  $A_p A_n^{-1}$  ratio of 2.1 for  $\mu\text{TEC}_A$  and 1.9 for  $\mu\text{TEC}_B$ . With further increased  $A_p A_n^{-1}$  ratio  $Q_{\text{max}}$  started decreasing again. The reduced electrical resistance with increasing  $A_p A_n^{-1}$  ratio lead to an increased optimal current and thus to an increased Peltier cooling, while the thermal conductance showed a linear increase. The slope of  $Q_{\text{max}}$  is a result of the almost logarithmically increasing cooling power, which was obtained by balancing the Joule heating, Peltier cooling, and thermal conduction, divided by the linearly increased top contact area. Therefore, choosing the optimal  $A_p A_n^{-1}$  ratio is again a trade-off between maximum net cooling temperature and  $Q_{\text{max}}$ . The  $A_p A_n^{-1}$  ratios 3.58 of  $\mu\text{TEC}_A$  and 2.64 of  $\mu\text{TEC}_B$  are marked in Figure 2d) with blue and red dashed lines, respectively. By decreasing the  $A_p A_n^{-1}$  ratio of  $\mu\text{TEC}_B$  from 3.58 to 2.64, the maximum net cooling temperature was increased by 17% from  $9.3$  to  $10.9 \text{ K}$  and  $Q_{\text{max}}$  increased from  $121.4 \text{ W cm}^{-2}$  to  $134.9 \text{ W cm}^{-2}$ . With the same  $A_p A_n^{-1}$  ratio as  $\mu\text{TEC}_B$ , the net cooling temperature of  $\mu\text{TEC}_A$  would be increased by 19% to  $7.6 \text{ K}$  combined with an increased  $Q_{\text{max}}$  of  $102 \text{ W cm}^{-2}$ .

### 3.2. Embedding the Devices within a Filling Material

Devices for use in commercial applications require a suitable packaging so that the devices sustain the direct thermal contact with the target heat sink and source. The as fabricated devices are free-standing, limiting their usage. Hence, we embedded the free-standing devices so that they build a flat surface at the top level. The embedding provides mechanical as well as chemical stability to the devices. With a filling material having a value of the thermal conductivity,  $\kappa_{\text{filling}}$ , the heat flux through the filling material provides a thermal shortcut. Consequently, the lower  $\kappa_{\text{filling}}$ , the better. We used a Finite Element Method (FEM) to select the filling material

and evaluated the quantitative effect of the material on the net cooling temperature of the  $\mu$ TEC. Whereas the device geometry and the electrical input current were kept constant,  $\kappa_{\text{filling}}$  was varied over five orders of magnitude, from  $0.001 \text{ W m}^{-1} \text{ K}^{-1}$  to  $100 \text{ W m}^{-1} \text{ K}^{-1}$ , keeping  $I$  at 180 mA. The simulation revealed that the maximum net cooling temperature decreased rapidly as  $\kappa_{\text{filling}}$  was increased (Section VI and Figure S6, Supporting Information). An increase in  $\kappa_{\text{filling}}$  from 0 to  $1 \text{ W m}^{-1} \text{ K}^{-1}$  led to a cooling of 2 K, from which point onward a further increase in the  $\kappa_{\text{filling}}$  led to a drastic reduction in the net cooling temperature. When  $\kappa_{\text{filling}}$  reached  $10 \text{ W m}^{-1} \text{ K}^{-1}$ , the corresponding cooling was 1.19 K. Almost no cooling was observed for a filling material with  $\kappa_{\text{filling}} = 100 \text{ W m}^{-1} \text{ K}^{-1}$ .

Based on the simulations, embedding with a polymer proved to be most effective because of its sufficiently low  $\kappa_{\text{filling}}$ , that is, instead of being free standing, the whole device was encapsulated leaving just the top Au open. We used photoresist (AZ9260), which has good processability and could be spin-coated with a thickness in the range of the device. Its viscosity enabled complete embedment of the device. Then we structured the photoresist to open the top Au portions and flood exposed the remaining parts such that the photoresist ended up on the same height as the device building a smooth surface at the top level, which is important for many applications. Importantly, the baking parameters of the photoresist did not affect the device properties negatively. The  $\kappa_{\text{filling}}$  of the photoresist at room temperature was  $0.15 \text{ W m}^{-1} \text{ K}^{-1}$ ,<sup>[43]</sup> which was used in the simulation of embedded  $\mu$ TECs (e $\mu$ TECs). The fabricated  $\mu$ TECs were used to study the effect of embedding with a filling material. The cooling characteristics with respect to the electric current were studied for embedded  $\mu$ TEC\_A (e $\mu$ TEC\_A) and  $\mu$ TEC\_B (e $\mu$ TEC\_B). The maximum cooling was obtained at the same value of electric current but the maximum cooling temperature of e $\mu$ TEC\_A and e $\mu$ TEC\_B was lower (Figure 3a). Adding the filling material reduced the overall cooling from 6.4 to 5.5 K for  $\mu$ TEC\_A and from 10.8 to 9.6 K for  $\mu$ TEC\_B. Notably, the shape of the cooling curve remained the same.

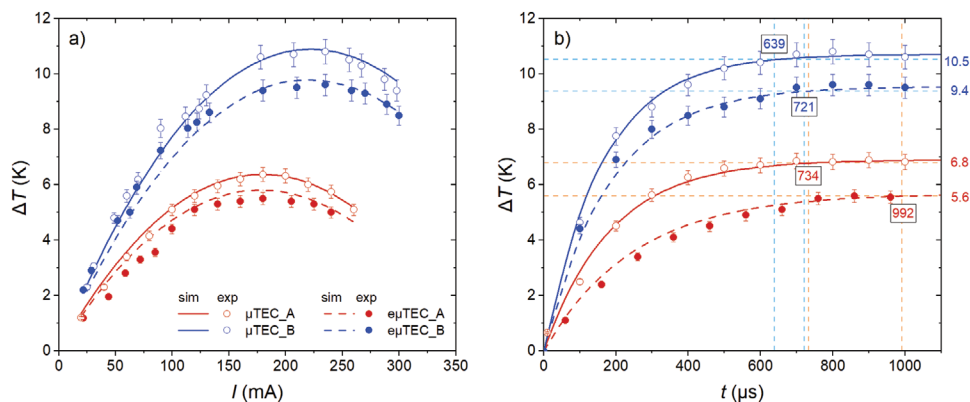
This reduction in cooling is evident from the simulation results (Figure S6a,b, Supporting Information) and the thermal image (Figure S6c, Supporting Information). The cooling curve of the embedded  $\mu$ TECs was simulated by using

the material properties from Table 1 and a contact resistance of  $5 \times 10^{-7} \Omega \text{ cm}^2$  for e $\mu$ TEC\_A and of  $1.6 \times 10^{-7} \Omega \text{ cm}^2$  for e $\mu$ TEC\_B. Good agreement between the simulation and experimental data was obtained for e $\mu$ TEC\_A and e $\mu$ TEC\_B when  $\kappa$  of the filling material was  $0.15 \text{ W m}^{-1} \text{ K}^{-1}$ . Thermal data reveal no signal around  $\mu$ TEC\_A but some heat dissipation from e $\mu$ TEC\_A to the filling material (Figure S6c, Supporting Information). This explains the slight degradation of the cooling performance of embedded devices.

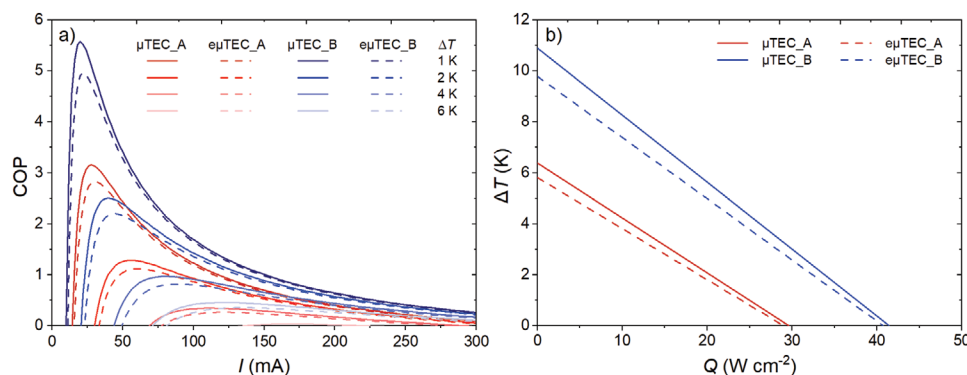
Embedding devices in the packaging material increases their thermal mass, which might lead to a reduced response time. To analyze the transient thermal response of the device a pulsed electric current was applied and the net cooling temperature was measured at different time delays using thermoreflectance microscopy (Figure 3b). The time dependent increase of the net cooling temperature was fitted (Figure 3b solid and dashed lines) by  $\Delta T = \Delta T_{\text{max}} (1 - \exp(-t/\tau))$ , where  $\Delta T_{\text{max}}$  is the steady-state maximum net cooling temperature, and  $\tau$  the thermal response time of the device. In the following we defined  $4\tau$  as thermal response time to reach the steady-state cooling temperature (98.2% of  $\Delta T_{\text{max}}$ ). The steady-state cooling temperature was reached at 734  $\mu\text{s}$  for  $\mu$ TEC\_A and 639  $\mu\text{s}$  for  $\mu$ TEC\_B, without filling material. The lower thermal response time of  $\mu$ TEC\_B is due to the reduced cross-section of the device. For the polymer-embedded devices, the thermal response time to reach steady-state has been increased by 35% to 992  $\mu\text{s}$  for e $\mu$ TEC\_A and by 13% to 721  $\mu\text{s}$  for e $\mu$ TEC\_B due to the additional thermal mass of the filling material.

### 3.3. Performance of the Embedded Devices

The efficiency of a TEC is typically described by the coefficient of performance (COP) given by the ratio of the absorbed heat at the cold side to the electrical input power. In order to analyze the efficiency of our  $\mu$ TECs and the effect of embedding, we simulated the COP of the devices. Details of the simulation can be found in the supporting information (Sections VII, VIII and Figures S7–S9, Supporting Information). Figure 4a shows the COP of  $\mu$ TEC\_A and  $\mu$ TEC\_B compared to the embedded  $\mu$ TECs e $\mu$ TEC\_A and e $\mu$ TEC\_B, simulated for a constant



**Figure 3.** Cooling performance of free-standing and embedded  $\mu$ TECs. a) Shows the simulation results (lines) and the experimental (dots) cooling response of e $\mu$ TEC\_A e $\mu$ TEC\_B in comparison to  $\mu$ TEC\_A,  $\mu$ TEC\_B; b) The comparison of the transient response time of free-standing and embedded  $\mu$ TECs at different time delays.



**Figure 4.** Realizing high Coefficient of Performance and Cooling power density of as fabricated devices. a) The effect of current on COP of freestanding and embedded devices ( $\mu\text{TEC}_A$ ,  $e\mu\text{TEC}_A$ ,  $\mu\text{TEC}_B$ , and  $e\mu\text{TEC}_B$ ). b) Comparison of cooling power density and net cooling temperature of freestanding and embedded devices ( $\mu\text{TEC}_A$ ,  $e\mu\text{TEC}_A$ ,  $\mu\text{TEC}_B$ , and  $e\mu\text{TEC}_B$ ).

temperature difference of 1, 2, 4, and 6 K in dependency of the applied current. To maintain a constant temperature difference, the absorbed heat at the cold side was adjusted depending on the applied current (Figure S8, Supporting Information). The maximum COP of 3.2 for  $\mu\text{TEC}_A$  and 5.6 for  $\mu\text{TEC}_B$  was found at the lowest temperature difference of 1 K. The maximum COP of the embedded devices decreased for  $e\mu\text{TEC}_A$  and  $e\mu\text{TEC}_B$  to 2.8 and 4.9, respectively, due to the additional heat losses caused by the filling material. In addition, as the temperature difference increased, the heat absorbed on the cold side decreased, resulting in a reduced COP, and the current at which the maximum COP was observed increased linearly.

Figure 4b shows the dependence of the net cooling temperature on the simulated cooling power density  $Q$  for the freestanding and embedded  $\mu\text{TECs}$ . Note that the  $Q$  shown in Figure 4b is the cooling power density of the device, that is, the distance of 50  $\mu\text{m}$  between the adjacent leg pairs was considered, which gives a filling factor of 23.9% of  $\mu\text{TEC}_A$  and 21.7% for  $\mu\text{TEC}_B$ .

The maximum net cooling temperature of  $\mu\text{TEC}_A$  was 6.4 K and of  $\mu\text{TEC}_B$  10.8 K. The maximum  $Q$  at zero temperature difference was  $29.6 \text{ W cm}^{-2}$  for  $\mu\text{TEC}_A$  and  $41.4 \text{ W cm}^{-2}$  for  $\mu\text{TEC}_B$ . Due to the additional heat losses when embedding the devices, the maximum net cooling temperature of  $\mu\text{TEC}_A$  and  $\mu\text{TEC}_B$  was reduced to 5.5 and 9.6 K, respectively. The maximum  $Q$  also showed slight decrease to  $29 \text{ W cm}^{-2}$  for  $\mu\text{TEC}_A$  and  $40.75 \text{ W cm}^{-2}$  for  $\mu\text{TEC}_B$ . By further reducing the distance between the leg pairs the maximum  $Q$  could be increased, that is, if the distance between the leg pair is 20  $\mu\text{m}$  (Section IX and Figure S10, Supporting Information) the maximum  $Q$  would increase to  $89.2 \text{ W cm}^{-2}$ , which is more than double the cooling power density of  $\mu\text{TEC}_B$ .

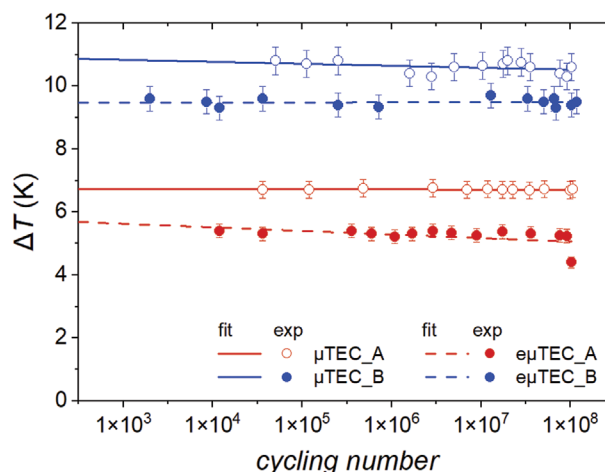
### 3.4. Reliability of the Embedded Devices

Previous studies have demonstrated the exceptional reliability of the free-standing  $\mu\text{TECs}$ .<sup>[1]</sup> The  $\mu\text{TEC}$  survived for 1 month in a cooling stability experiment under the continuous application of a direct current. In the cycling reliability test, the device started to deteriorate after 10 million cycles. Since practical

applications require that the devices be embedded, we explored how this embedding affects the reliability of different  $\mu\text{TECs}$  (Figure 5).

To characterize the cycling reliability, we applied a pulsed electrical current (1 ms on and 4 ms off) to  $\mu\text{TEC}_A$  for 120 million cycles. The net cooling temperature, which was measured by thermoreflectance microscopy, was seen to remain intact even after 120 million cycles.  $\mu\text{TEC}_B$  also showed a high cycling reliability of over 70 million cycles.

Optimizing the geometry of the  $\mu\text{TEC}$  led to an exceptionally high cycling reliability. In the previous work, the deterioration in reliability was most likely because of a loose contact between the thermoelectric leg and the top metallization. In this study, we increased the contact area between these parts and achieved a higher cycling reliability for the  $\mu\text{TECs}$ . The cycling reliability was also tested for the polymer-embedded  $\mu\text{TECs}$ .  $e\mu\text{TEC}_A$  was found to survive 85 million cycles, whereas  $e\mu\text{TEC}_B$  survived 70 million cycles, following which all two devices showed a drop in the net cooling temperature. It is apparent that the polymer-embedded  $\mu\text{TEC}$  was less tolerant to the internal strain caused by the materials' thermal expansion coefficients than



**Figure 5.** Comparison of cycling reliability of the free-standing and embedded  $\mu\text{TECs}$ . One cycle corresponds to an electric pulsed current with an on time of 1 ms and off time of 4 ms.



the free-standing  $\mu$ TEC. This is because the free-standing  $\mu$ TEC had more freedom to adapt to this internal strain, whereas the polymer-embedded  $\mu$ TEC had less breathing space to cope with the internal strain. Nevertheless, before it started to deteriorate the cycling performance of the embedded  $\mu$ TEC was exceptionally high. Such performance is more than sufficient for most applications.

#### 4. Conclusion

We combined photolithographic patterning and electrochemical deposition to fabricate free-standing  $\mu$ TECs and then compared the net cooling temperature of two  $\mu$ TECs with different geometries. We found that the best performing device had a reduced ratio between n-type and p-type cross-sections of the thermoelectric legs closer to the optimal ratio, increased leg height, and decreased contact resistance. At room temperature, this  $\mu$ TEC yielded a net cooling temperature of 10.8 K at 235 mA. The device showed a rapid cooling-response time of 639  $\mu$ s and a cycling reliability of 120 million cycles. The effect of geometry and contact resistance on the cooling power density and the net cooling temperature was studied, giving a guideline for the fabrication of TECs with optimal cooling power density and net cooling temperature. Real-world applications require that the  $\mu$ TECs be packaged so that they can be brought in direct thermal contact with the heat sink. We thus embedded the  $\mu$ TECs with the polymer AZ9260 photoresist and characterized the performance of the  $\mu$ TECs before and after embedding. The net cooling temperature after embedding within the polymer was 9.6 K at 235 mA, a drop that could be attributed to thermal loss through the polymer. However, the embedded devices showed a rapid cooling response time of 721  $\mu$ s and a slightly reduced cycling reliability of 85 million cycles. These embedded, optimized, stable and easily scalable  $\mu$ TECs open new avenues for wide-spread applications in biomedical applications, powering internet-of-things devices, and local heat management.

#### Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

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#### Conflict of Interest

The authors declare no conflict of interest.

#### Data Availability Statement

The data that support the findings of this study are available from the corresponding author upon reasonable request.

#### Keywords

contact resistance, electrochemical deposition, micro-thermoelectric devices, packaging, TEC design, TEC reliability, thermoelectrics

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