



Tuning the magnetic properties of permalloy-based magnetoplasmonic crystals for sensor applications

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Miniature magnetic sensors based on magnetoplasmonic crystals (MPICs) exhibit high sensitivity and high spatial resolution, which can be obtained by the excitation of surface plasmon polaritons. The field dependence of surface plasmon polaritons' enhanced magneto-optical response strongly correlates with the magnetic properties of MPICs that can be tuned by changing spatial parameters, such as the period and height of diffraction gratings and thicknesses of functional layers. This work compares the magnetic properties of MPICs based on Ni₈₀Fe₂₀ (permalloy) obtained from local (longitudinal magneto-optical Kerr effect) and bulk (vibrating-sample magnetometry) measurements and demonstrates an ability to control sensors' performance through changing the magnetic properties of MPICs. The influence of the substrate's geometry (planar or sinusoidal and trapezoidal diffraction grating profiles) and the thickness of the surface layer is examined.

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1. Introduction

Recent rapid advancements in nanomaterial development have had a huge impact on the elaboration of different devices, including magnetic field sensors. Nowadays, magnetic field sensors are being used in orientation devices,^{1,2)} control^{3,4)} and security^{5,6)} systems, medicine,^{7,8)} space exploration^{9,10)} and other noteworthy applications. Depending on the application, magnetic field sensors have different linear dimensions and levels of accuracy, efficiency, locality and sensitivity. However, the modern tendency for miniaturisation of devices has led to a stronger necessity for highly localised and sensitive magnetic field sensors.

Nowadays, the most sensitive and widely used magnetic field sensors are based on the Hall effect^{11,12)} and on superconducting quantum interference devices.^{13,14)} However, the sensitivity of these devices drops with a decrease of the probe area.^{15,16)} Another possible way to design magnetic field sensors neglecting the influence of linear dimensions on the sensitivity and locality of the device is based on the use of magneto-optical effects in Voigt geometry.^{17–19)} The magnitude of magneto-optical response is usually very weak even for ferromagnetic materials, but it can be enhanced by the use of magnetoplasmonic crystals (MPICs).^{20,21)} MPICs are periodically nanostructured ferromagnetic surfaces^{22,23)} that, due to diffraction effects, support the excitation and propagation of surface resonant evanescent coupled oscillations of metallic plasma and photons called surface plasmon polaritons.^{24,25)} This method allows one to measure precisely the external magnetic field in the area limited only by the size of the optical spot and the penetration depth of light. Recently, a prototype of permalloy-based magnetic field sensor has achieved sensitivity as high as 4.4×10^{-6} Oe in a modulating alternating field of 5.0 Oe at an area of 1 mm.²⁶⁾ According to theoretical calculations, its sensitivity can be tuned up to 10^{-7} Oe²⁷⁾ which is appropriate for biomedical applications.

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It has been proved that the magneto-optical response of MPIC structures strongly correlates with their magnetic properties, which in turn allows the tuning of the magnetic field

sensor's parameters.^{26,28,29)} This fact opens up a way to improve the sensitivity of MPIC-based magnetic field sensors by noise reduction and amplification of magneto-optical response by varying the MPIC's parameters such as the ferromagnetic layer thickness and the diffraction grating period and height.

In our previous work,²⁶⁾ estimations of the signal-to-noise ratio were done to calculate sensitivity, the measurable field region and the required modulating AC magnetic field for sensing applications. It was found that an increase of the ferromagnetic layer thickness leads to the occurrence of a stepped behaviour of the field-dependent magneto-optical response and the decrease of signal-to-noise ratio values. In this paper, an explanation of such stepped behaviour is shown. A way to tune the coercive force and effective anisotropy of MPICs by changes in the ferromagnetic layer thickness and spatial parameters of the substrates is revealed.

2. Experimental methods

MPICs were fabricated using the DC magnetron sputtering technique. Two series of samples were made with the deposition of 100 nm or 130 nm permalloy (Ni₈₀Fe₂₀ or Py) onto smooth planar silicon substrates, Sub₁, and polymer substrates, Sub₂ and Sub₃, with quasi-sinusoidal and trapezoidal diffraction gratings, respectively. The spatial parameters of the gratings are as follows: periods $d_2 = 320$ nm and $d_3 = 740$ nm and profile heights $h_2 = 20$ nm and $h_3 = 100$ nm for Sub₂ and Sub₃ respectively. The deposition process was carried out at a temperature below 100 °C, constant pressure of 3 mTorr, constant argon flow equal to 10 ccm and constant power of 50 W, to prevent heat deformation of the polymer substrates. Images of the fabricated MPICs' surfaces based on Sub₂ and Sub₃ were obtained by an atomic force microscope, and are shown in Fig. 1. The obtained images show that the nanostructures' height and periodicity errors were in the vicinity of 5%. This MPICs were found to have a strong in-plane anisotropy that formed an easy magnetisation axis (EMA), directed along the stripes of the diffraction gratings, and a hard

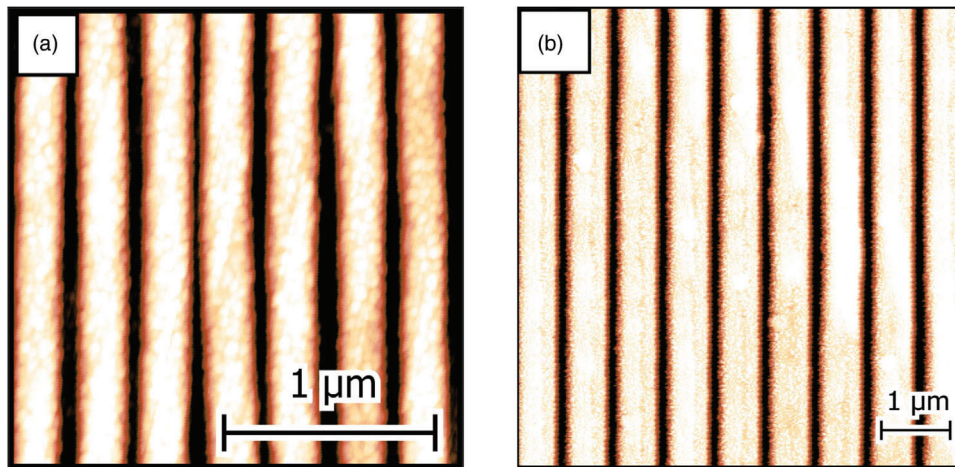


Fig. 1. (Colour online) Atomic force microscope images of MPICs' surfaces based on a) Sub₂ and b) Sub₃.

magnetisation axis, perpendicular to the stripes.^{30,31)} Also, the MPICs based on the Sub₂ and Sub₃ substrates had different anisotropy coefficients in regions with tilted and bent parts due to their morphological features.³²⁾ The films deposited onto Sub₁ were magnetically isotropic.^{30,31)} Measurements of the integral and local magnetic properties in an external magnetic field along the EMA were done with a vibrating-sample magnetometer (VSM) and with a NanoMOKE III magneto-optical magnetometer in the longitudinal magneto-optical Kerr effect (LMOKE) geometry,

respectively. Also, the integral magnetic properties for the Sub₃-based MPICs were measured in-plane along the EMA, out-of-plane and additionally at 45° relative to the plane. All measurements were done at room temperature. To analyse the magnetisation reversal processes of the MPICs, the differential susceptibility was calculated as the first-order derivative of the magnetisation with respect to the applied field in hysteresis loops, $\chi_d = dM/dH$. χ_d also represents the switching field distribution (SFD). The FWHM for the calculated $\chi_d/\chi_d(\max)$ was estimated to determine the

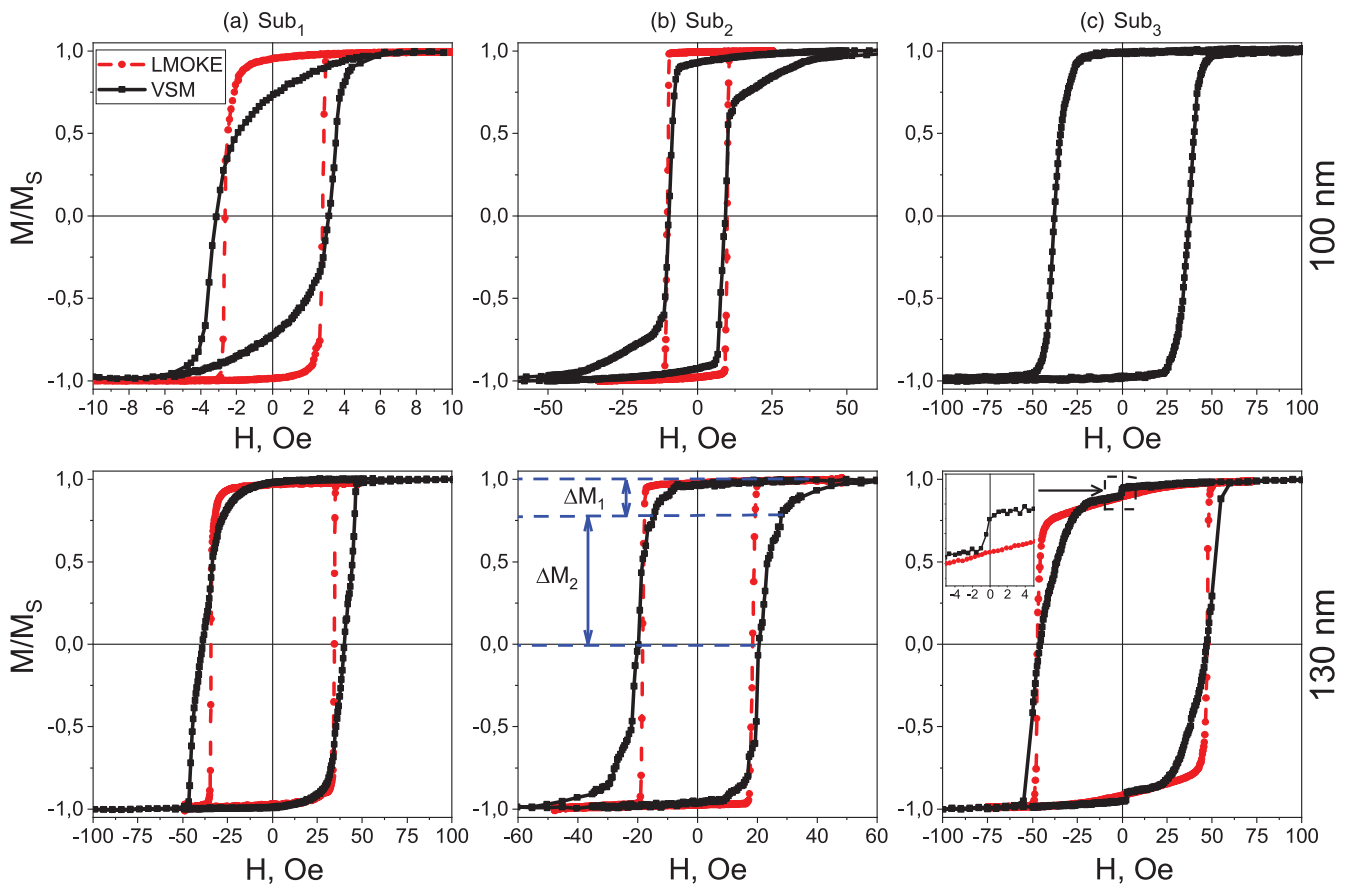


Fig. 2. (Colour online) Hysteresis loops along EMA for MPICs with different thicknesses of Py layer obtained by integral (black solid line with rectangles) and local (red dashed line with circles) methods. The first and second rows represent the hysteresis loops for MPICs with permalloy thicknesses of 100 and 130 nm, respectively. Columns (a)–(c) represent data for the MPICs based on Sub₁, Sub₂ and Sub₃, respectively. ΔM_1 is the change of magnetisation associated with the magnetisation vector rotation and ΔM_2 is the change of magnetisation related to the domain wall propagation.

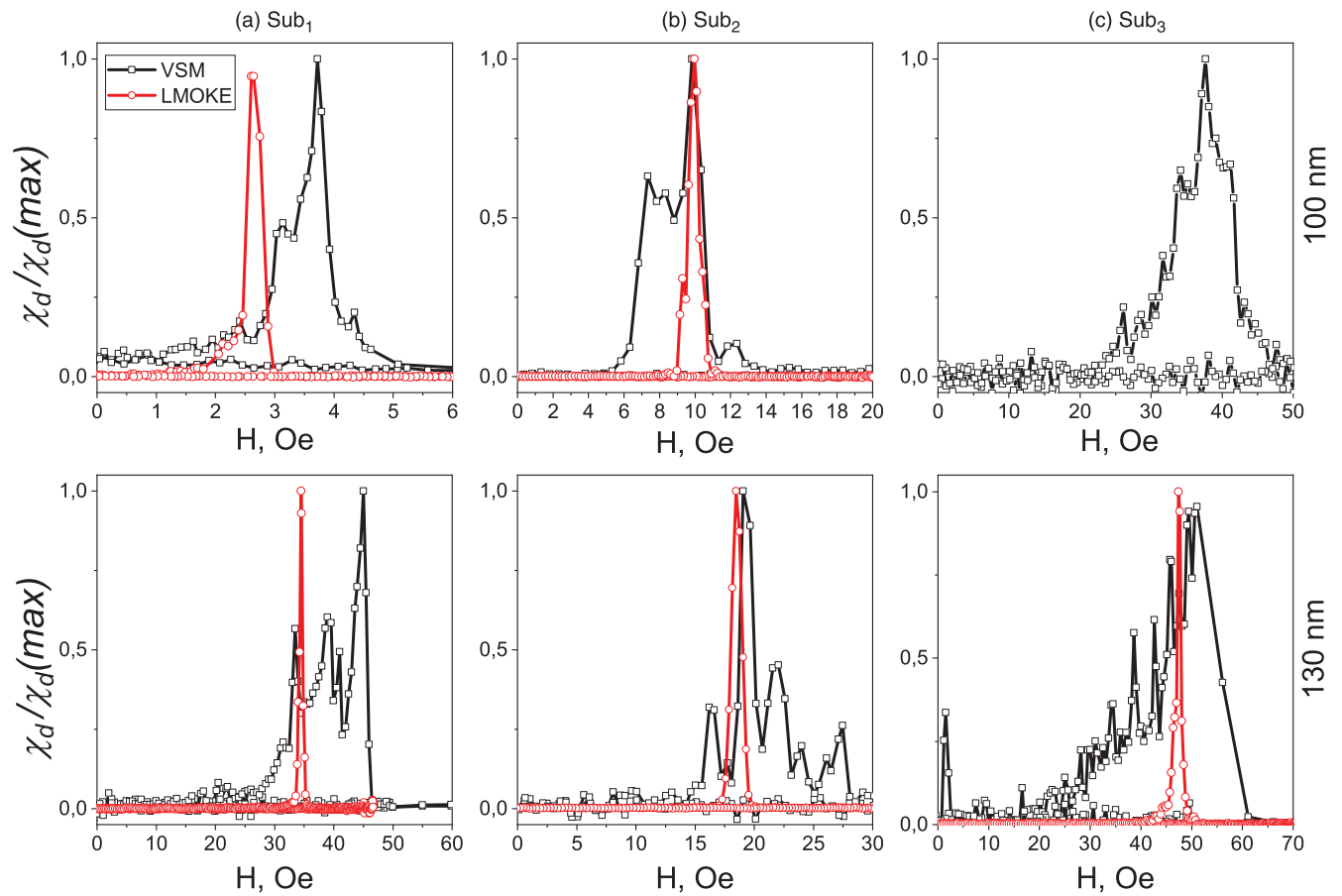


Fig. 3. (Colour online) Differential susceptibility normalised to the maximum $\chi_d/\chi_d(\max)$ for MPICs with different Py thicknesses obtained by integral (black solid line with open rectangles) and local (red lines with open circles) methods. The first and the second rows correspond to the samples with permalloy layers of 100 and 130 nm, respectively. Columns (a)–(c) represent data for the MPICs based on Sub₁, Sub₂ and Sub₃, respectively.

contributions of morphology to the magnetisation reversal of the MPICs.³³⁾

3. Results and discussion

3.1. Results

The dependences of the magnetic moment normalised to a saturation value on the external magnetic field measured by local and integral methods are shown in Fig. 2. The methods used to measure the magnetic properties have a principal difference arising from their features: the VSM allows measurement of the magnetic moment of the whole volume of the sample, including defects at its edges caused by the fabrication process;²⁸⁾ on the other hand, the LMOKE geometry allows examination of the local magnetic properties, in an area limited by the light penetration depth and the diameter of the light beam.

A rectangular hysteresis loop of the surface area means the magnetisation reversal was mainly driven by domain wall propagation in regions far from the edges. Magnetisation reversal proceeded additionally via rotation of the magnetisation vector and led to the tilting of the hysteresis loops in all cases for the VSM measurements. As a result, the integral hysteresis loops presented a smaller squareness ratio $Q = M_r/M_S$ (M_r being the remanent magnetisation and M_S the saturation magnetisation) than the local ones and the differential susceptibility calculated for the LMOKE loops had a narrower peak than that for the VSM.^{34,35)} The dependences of χ_d normalised to the maximum value on

the magnetic field are shown in Fig. 3. The broadening of the SFD peaks for the VSM measurements (Fig. 3) may be connected with one of the following factors: (i) the polycrystalline nature of the permalloy films with a finite number of large grains of the structure; (ii) magnetisation reversal processes of edge defects themselves; (iii) magnetic anisotropy fields having different magnitudes of tilted and bent areas on the diffraction grating; and (iv) thickness-dependent mechanical stress in the permalloy layer.

The hysteresis loops measured with the VSM and LMOKE were only similar at fields around the coercive force H_C for each sample. Taking into account the ratio of the regions where the magnetisation was associated with the magnetisation vector rotation (ΔM_1 in Fig. 2) and was related to the domain wall propagation (ΔM_2 in Fig. 2), one can see that the main volume of the samples was magnetised through the domain wall propagation. The increase of in-plane H_C along the EMA with the permalloy layer thickness for both types of measurements is in agreement with theory and can be explained by the broadening of the classical Bloch wall with the increase of permalloy thickness from 100 nm to 130 nm.³³⁾

The positions of the χ_d peaks were almost the same for the VSM and LMOKE loops for the MPICs based on Sub₂ and Sub₃, while for the samples based on the smooth Sub₁ substrates the χ_d peaks' positions shifted towards lower fields for the LMOKE curves. This can be explained by the existence of geometry-driven anisotropy dominating in the

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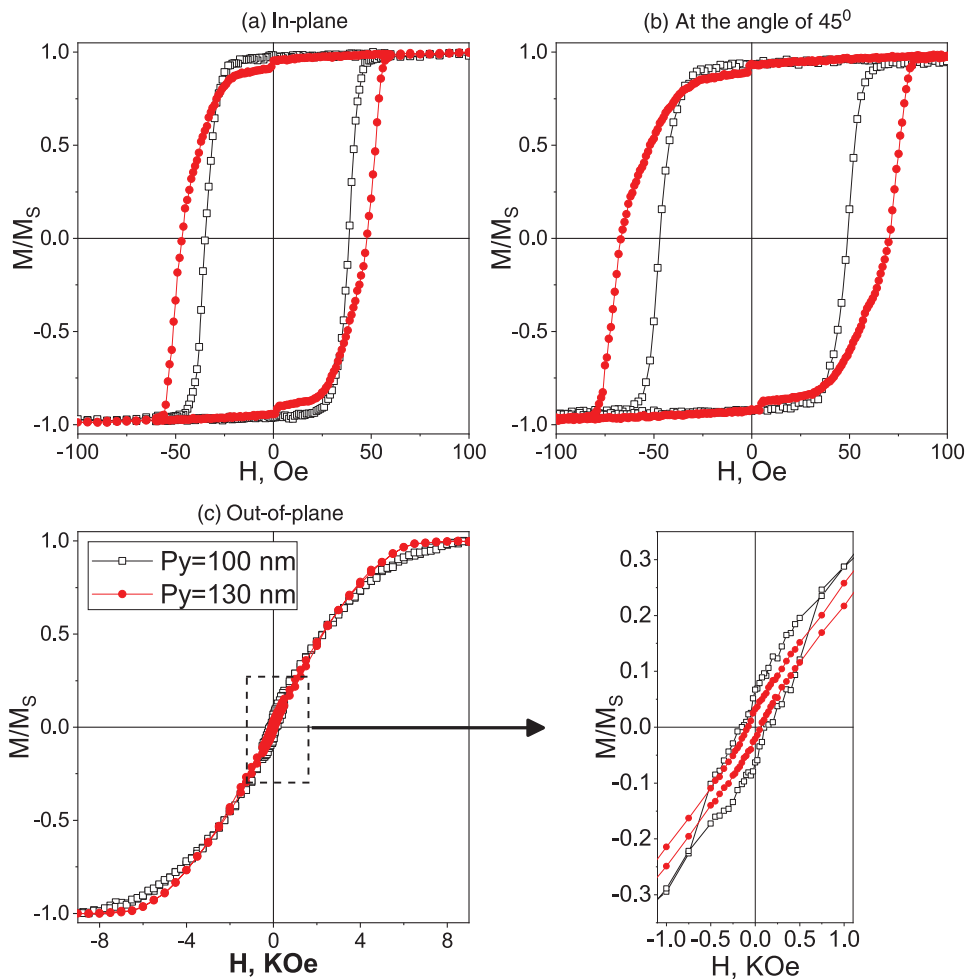


Fig. 4. (Colour online) Hysteresis loops of MPICs with permalloy thickness of 100 nm (black solid line with open rectangles) and 130 nm (red solid lines with solid circles) based on Sub₃. Panels (a)–(c) correspond to the loops measured with the VSM in-plane along the EMA, under the angle of 45° relative to the plane and in out-of-plane directions of the applied field, respectively.

MPICs and a stress effect through the thickness of the polycrystalline substrate-constrained films leading to the decrease of the anisotropy constant in the stress-free surface area.

Considering the VSM measurements for the smooth planar Sub₁ based MPICs, additional peaks did not repeat in the magnitude, field position or number of ascending and descending parts of the hysteresis loops. This result is associated with the noise level of the system. On the other hand, the position for some peaks in the χ_d functions for MPICs based on the Sub₂ and Sub₃ substrates having morphological features repeated but the noise level also remained very high. The increase of permalloy layer thickness led to the heterogeneous behaviour of magnetic anisotropy and χ_d , resulting in the appearance of additional peaks for the MPIC with a 130-nm-thick permalloy layer in comparison with MPICs based on thinner films. To determine the origin of all the peaks, additional structural and magnetic measurements should be done, for example, by the use of the first-order reversal curve (FORC) method.^{36,37)}

In our previous work, devoted to the application of the considered MPICs as magnetic field sensors, it was shown that the required external modulation field and signal-to-noise ratio depend on the magnetic properties of the MPICs.²⁶⁾ In particular, the required external modulation field is related to the coercive force, H_C , and a measurable field region is

determined by the slope of the signal-to-noise ratio function that has a phenomenological connection with the magnetisation curves. More information can be found in Ref. 26. The additional peaks at 46 Oe and 50 Oe seen in the χ_d function for the MPIC with a permalloy thickness of 130 nm based on Sub₃ [Fig. 2(c) bottom, VSM] correspond to regions of steps on a step-like field dependence of the signal-to-noise ratio function. There was an ultra-soft repeatable magnetic phase with an H_C value of 1.5 Oe. This magnetic phase in the future could be used to measure external magnetic fields in very low modulating fields with magnitudes of less than 1 Oe. The contribution of the ultra-soft magnetic phase can be further increased by tuning the ferromagnetic layer thickness.

Additionally, the hysteresis loops for the Sub₃-based MPICs were measured with a VSM in three directions of the external magnetic field. The measured loops are shown in Fig. 4. The sequential change of the magnetisation direction from in-plane along the EMA to an angle of 45° relative to the plane, and then to the out-of-plane direction, led to the increase of H_C . The mechanism of this effect is connected to the presence of some additional EMAs out of the sample's plane. They can be associated with morphological features that appeared with the changing of the magnetisation direction³²⁾ of the MPIC based on the Sub₃ substrate.

To estimate the influence of the MPICs' morphology on the magnetisation reversal process the FWHM of the main

Table I. FWHM of $\chi_d/\chi_d(\text{max})$, measured by VSM and LMOKE techniques.

Measurement technique	Substrate//Thickness of Py, nm					
	Sub ₁ //100	Sub ₁ //130	Sub ₂ //100	Sub ₂ //130	Sub ₃ //100	Sub ₃ //130
VSM	0.52	2.44	3.35	4.42	8.40	11.10
LMOKE	0.24	0.6	0.87	1.02	0.69	0.82

peaks of the χ_d functions was calculated and the results are presented in Table I.

The appearance of a profile with a sinusoidal or trapezoidal shape (Sub₂ or Sub₃ substrate, respectively) was followed by the growth of $\chi_d/\chi_d(\text{max})$ width in comparison with the smooth Sub₁ based samples. Changes in the permalloy layer thickness also resulted in the broadening of the FWHM regardless of the measurement method for each sample. For the integral VSM measurements, one can see that the increase of the h and d parameters led to the broadening of $\chi_d/\chi_d(\text{max})$ width related to the prevailing contribution of magnetisation reversal processes (i)–(iv). On the other hand, for the local LMOKE measurements there was a decrease of $\chi_d/\chi_d(\text{max})$ width with increasing h and d parameters. This is presumably related to the surface microstructure of the MPICs being determined by the features of the permalloy grains (depending on the MPIC profile shape) rather than by stress-related factors from the substrate. Broadening means that there were larger grains in the permalloy film³⁸⁾ considering the MPICs based on Sub₂ in comparison with the Sub₃-based MPICs.

In summary, we have established that the step-like behaviour of magneto-optical properties²⁶⁾ correlates with the integral magnetic properties, but does not correlate with the local ones. This can be explained by the appearance of millimetre-scale features (region of surface for signal-to-noise ratio studies) and the absence of micron-scale features (LMOKE studies), which will be studied further.

4. Conclusions

The detailed measurements of magnetic properties provided in this paper show a way to tune the required modulation field and the measurable field region of a magnetic field sensor based on MPICs. This can be done by changing the ferromagnetic layer thickness and the substrate's parameters influencing the magnetic and magneto-optical properties. It was confirmed that the origin of the multi-region MPIC-based sensor is the additional magnetic phases in the MPIC based on Sub₃ with 130 nm permalloy. This makes it possible to create a magnetic field sensor working in two different regions of the measurable magnetic field. In future works, the contributions of the magnetisation vector rotation and domain wall propagation will be additionally studied by local and integral FORC diagrams in multiphase MPICs.

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