

## High-magnetic field characterization of magnetocaloric effect in FeZrB(Cu) amorphous ribbons

P. Alvarez-Alonso, J. L. Sánchez Llamazares, C. F. Sánchez-Valdés, M. L. Fdez-Gubieda, Pedro Gorria, and J. A. Blanco

Citation: *Journal of Applied Physics* **117**, 17A710 (2015); doi: 10.1063/1.4907188

View online: <http://dx.doi.org/10.1063/1.4907188>

View Table of Contents: <http://scitation.aip.org/content/aip/journal/jap/117/17?ver=pdfcov>

Published by the AIP Publishing

---

### Articles you may be interested in

[Magnetocaloric effect in Fe-Zr-B-M \(M=Ni, Co, Al, and Ti\) amorphous alloys](#)

*J. Appl. Phys.* **116**, 093910 (2014); 10.1063/1.4895048

[Enhanced refrigerant capacity and magnetic entropy flattening using a two-amorphous FeZrB\(Cu\) composite](#)

*Appl. Phys. Lett.* **99**, 232501 (2011); 10.1063/1.3665941

[Amorphous-FeCoCrZrB ferromagnets for use as high-temperature magnetic refrigerants](#)

*J. Appl. Phys.* **99**, 08K909 (2006); 10.1063/1.2172234

[Characterization of amorphous FeZrB\(Cu\) alloys by the inductance spectroscopy method](#)

*J. Appl. Phys.* **87**, 7112 (2000); 10.1063/1.372947

[Effects of Co addition on magnetic properties and nanocrystallization in amorphous Fe<sub>84</sub>Zr<sub>3.5</sub>Nb<sub>3.5</sub>B<sub>8</sub>Cu<sub>1</sub> alloy](#)

*J. Appl. Phys.* **86**, 6301 (1999); 10.1063/1.371690

---

The advertisement features a blue background with a glowing light effect on the right side. On the left, there is a small image of the 'AIP Applied Physics Reviews' journal cover, which shows a 3D grid structure and a graph. The main text 'NEW Special Topic Sections' is written in large, white, bold letters. Below this, the text 'NOW ONLINE' is in yellow, followed by 'Lithium Niobate Properties and Applications: Reviews of Emerging Trends' in white. The AIP Applied Physics Reviews logo is in the bottom right corner.

**NEW Special Topic Sections**

**NOW ONLINE**  
Lithium Niobate Properties and Applications:  
Reviews of Emerging Trends

**AIP** Applied Physics  
Reviews

# High-magnetic field characterization of magnetocaloric effect in FeZrB(Cu) amorphous ribbons

P. Alvarez-Alonso,<sup>1,a)</sup> J. L. Sánchez Llamazares,<sup>2</sup> C. F. Sánchez-Valdés,<sup>2</sup>  
 M. L. Fdez-Gubieda,<sup>1,3</sup> Pedro Gorria,<sup>4</sup> and J. A. Blanco<sup>5</sup>

<sup>1</sup>Departamento de Electricidad y Electrónica, Universidad del País Vasco (UPV/EHU), 48940 Leioa, Spain

<sup>2</sup>Instituto Potosino de Investigación Científica y Tecnológica A.C., Camino a la Presa San José 2055 Col. Lomas 4<sup>a</sup>, San Luis Potosí, S.L.P. 78216, Mexico

<sup>3</sup>BCMaterials, Edificio 500, Parque Científico y Tecnológico de Zamudio, 48160 Derio, Spain

<sup>4</sup>Departamento de Física & IUTA, EPI, Universidad de Oviedo, 33203 Gijón, Spain

<sup>5</sup>Departamento de Física, Universidad de Oviedo, Calvo Sotelo st. s/n, 33007 Oviedo, Spain

(Presented 6 November 2014; received 22 September 2014; accepted 16 October 2014; published online 6 February 2015)

The magnetic and magnetocaloric properties of a series of Fe-rich FeZrB(Cu) amorphous ribbons were investigated under magnetic field values up to  $\mu_0 H$  of 8 T. A correlation between the saturation magnetization and the maximum magnetic entropy change  $|\Delta S_M^{\text{peak}}|$  is clearly evidenced. Although these metallic glasses show relatively low  $|\Delta S_M^{\text{peak}}|$  values (from 3.6 to 4.4 J kg<sup>-1</sup> K<sup>-1</sup> for  $\mu_0 \Delta H = 8$  T), the  $\Delta S_M(T)$  curve broadens upon the increase in  $\mu_0 \Delta H$ , giving rise to a large refrigerant capacity  $RC$  (above 900 J kg<sup>-1</sup> for  $\mu_0 \Delta H = 8$  T). Using the universal curve method for rescaling the  $\Delta S_M(T, \mu_0 \Delta H)$  curves, we found a collapse of the curves around the Curie temperature. However, in the low-temperature range the curves do not match into a single one due to the existence of magnetic frustration. © 2015 AIP Publishing LLC.

[<http://dx.doi.org/10.1063/1.4907188>]

## INTRODUCTION

Fe-rich FeZrB(Cu) metallic glasses display striking magnetic behaviors such as low-temperature magnetic frustration and magneto-volume anomalies below the Curie temperature,  $T_C$ , due to the strong competition between Fe-Fe magnetic interactions.<sup>1-4</sup> Furthermore, the value of  $T_C$  scales almost linearly with the Fe content in the 210–350 K range,<sup>4,5</sup> and can be easily selected by changing the composition. The latter is of particular interest in magnetocaloric (MC) materials for: (a) developing active magnetic regenerators operating around room temperature<sup>6</sup> and (b) designing two-phase MC composites with a table-like shape of the  $\Delta S_M(T)$  curve. This usually improves the refrigerant capacity,  $RC$ ,<sup>7,8</sup> an important figure of merit that measures the transferable heat in an ideal thermodynamic cycle from the cold to the hot reservoirs at temperatures  $T_{\text{cold}}$  and  $T_{\text{hot}}$ , respectively.<sup>9,10</sup>

However, for these purposes, the materials must fulfill other requirements: to exhibit analogous values for the maximum magnetic entropy change,  $|\Delta S_M^{\text{peak}}|$ , and also similar shaped  $\Delta S_M(T)$  curves. Previous studies reported that for  $\mu_0 \Delta H \leq 5$  T, the FeZrBCu ribbons satisfy both conditions.<sup>11</sup> Moreover, the similarity of their  $\Delta S_M(T)$  curves was verified in the framework of the so-called “universal curve.”<sup>11</sup> In this way, it was shown theoretically, and verified phenomenologically for  $\mu_0 \Delta H$  up to 5 T, that the  $\Delta S_M(T, \mu_0 \Delta H)$  curves of different materials with similar critical exponents collapse into a single curve when properly rescaled.<sup>12-14</sup> The so-

called “universal curve” serves to extrapolate the magnetic entropy change to higher  $\mu_0 \Delta H$  values and wider temperature ranges,<sup>15</sup> and to reveal the presence of magnetic inhomogeneities.<sup>11</sup>

In this study, we investigate the magnetic and magnetocaloric properties of several Fe-rich FeZrB(Cu) amorphous ribbons and test the validity of the phenomenological universal curve up to high values of the magnetic field change  $\mu_0 \Delta H$  of 8 T.

## EXPERIMENTAL PROCEDURE

Seven amorphous ribbons with chemical composition given by: Fe<sub>90</sub>Zr<sub>10</sub>, Fe<sub>90</sub>Zr<sub>9</sub>B<sub>1</sub>, Fe<sub>91</sub>Zr<sub>7</sub>B<sub>2</sub>, Fe<sub>90</sub>Zr<sub>8</sub>B<sub>2</sub>, Fe<sub>88</sub>Zr<sub>8</sub>B<sub>4</sub>, Fe<sub>86</sub>Zr<sub>7</sub>B<sub>6</sub>Cu<sub>1</sub>, and Fe<sub>87</sub>Zr<sub>6</sub>B<sub>6</sub>Cu<sub>1</sub> were fabricated by melt spinning from arc melted bulk alloys. The amorphous character of the samples was confirmed by X-ray diffraction (no traces of crystalline phases were detected).

Magnetization measurements were performed on a Quantum Design PPMS-9T platform by using the vibrating sample magnetometer option. The  $T_C$  of samples was determined from the temperature dependence of the magnetization,  $M(T)$ , measured under a low applied magnetic field  $\mu_0 H$  of 5 mT. The isothermal magnetization curves,  $M(\mu_0 H)$ , were measured up to  $\mu_0 H = 8$  T from 50 to 400 K with  $T$ -steps of 10 K. The  $\Delta S_M(T)$  curve for each sample was obtained by numerical integration of the Maxwell relation (i.e.,  $\Delta S_M(T, \mu_0 H) = \mu_0 \int_0^{\mu_0 H} \left( \frac{\partial M(T', \mu_0 H')}{\partial T'} \right)_{T'=T} dH'$ ).<sup>6</sup>

For a given value of the magnetic field change  $\mu_0 \Delta H$ ,  $RC$  can be estimated on a first approach as the product  $|\Delta S_M^{\text{peak}}| \times \delta T_{\text{FWHM}}$ .<sup>10</sup> In this definition,  $\delta T_{\text{FWHM}}$

<sup>a)</sup>Author to whom correspondence should be addressed. Electronic mail: pablo.alvarez@ehu.es.

corresponds to the full-width at half-maximum of the  $\Delta S_M(T)$  curve [i.e.,  $\delta T_{\text{FWHM}}(\mu_0\Delta H) = T_{\text{hot}} - T_{\text{cold}}$ , with  $T_{\text{hot}}$  and  $T_{\text{cold}}$  the working temperature ends of the refrigerant thermodynamic cycle].

The phenomenological universal curve was obtained by an appropriate renormalization of the  $\Delta S_M(T)$  curves determined for the various  $\mu_0\Delta H$  values. The procedure for each  $\mu_0\Delta H$  value is:<sup>16</sup> (a) the  $\Delta S_M(T)$  curves are normalized to  $|\Delta S_M^{\text{peak}}|$ ; (b) the temperature axis is rescaled below and above  $T_C$  by imposing two temperatures,  $Tr_1$  and  $Tr_2$ , related to two reference points at each side of the  $\Delta S_M(T)$  curve that corresponds to a certain fraction of  $|\Delta S_M^{\text{peak}}|$  (i.e.,  $a \times |\Delta S_M^{\text{peak}}|$ , with  $a$  an arbitrary value between 0 and 1)

$$\begin{aligned} \theta &= -(T - T_C)/(Tr_1 - T_C) & T < T_C, \\ \theta &= (T - T_C)/(Tr_2 - T_C) & T > T_C. \end{aligned} \quad (1)$$

It must be pointed out that these reference temperatures do not have a physical meaning due to the arbitrariness of  $a$ . However,  $a$  is generally chosen equal to 0.5 since it corresponds to the half maximum and, therefore,  $Tr_1$  and  $Tr_2$  correspond to  $T_{\text{cold}}$  and  $T_{\text{hot}}$ , respectively.

## RESULTS AND DISCUSSION

Fig. 1(a) shows the typical low-field  $M(T)$  curves measured on heating after a zero-field-cooling procedure. All the samples exhibit a broad second-order magnetic phase transition. The value of  $T_C$  was estimated from the minimum of the  $dM/dT(T)$  curve. As listed in Table I,  $T_C$  values range between 210 and 320 K. Fig. 1(c) depicts the curves at 50 K normalized to their respective value at  $\mu_0H = 8$  T. The magnetic anisotropy increases with the Fe-content; in fact, for Fe at. % > 88, the ribbons hardly reach the saturation state at 8 T. Such behavior arises from the magneto-volume instabilities of these metallic glasses.<sup>1,2</sup> We have estimated the saturation magnetization,  $M_S$ , of the ribbons by fitting the corresponding isothermal  $M(\mu_0H)$  curve [a typical set of these  $M(\mu_0H)$  curves is given in Fig. 1(b) for  $\text{Fe}_{91}\text{Zr}_7\text{B}_2$ ] using an approach-to-saturation law.<sup>17</sup> The temperature dependence of  $M_S$  (see inset in Fig. 1) shows that the higher the Fe content, the more pronounced the decrease of the  $M_S(T)$  curves and the lower the saturation magnetization.

The samples exhibit broad  $|\Delta S_M(T)|$  curves [see Fig. 2(a)] and moderate peak values for the magnetic entropy change,  $|\Delta S_M^{\text{peak}}|$ , in consonance with the wide magnetic phase transition observed in the  $M(T)$  curves [see Fig. 1(a)]. It is also worth noting that the temperature corresponding to  $|\Delta S_M^{\text{peak}}|$ ,  $T^{\text{peak}}$ , as well as the value of  $|\Delta S_M^{\text{peak}}|$  follows a linear dependence with the at. % of Fe [see Fig. 2(b)], likewise  $T_C$  (see Table I & Refs. 4 and 5). In Fig. 2(c), we show the  $|\Delta S_M^{\text{peak}}|$  vs. the saturation magnetization estimated at 50 K ( $M_{S,50}$ ) for  $\mu_0\Delta H = 2, 5$ , and 8 T. A linear correlation between both magnitudes (with a positive slope increasing from 0.012 up to 0.024 when  $\mu_0\Delta H$  increases from 2 to 8 T) is clearly evidenced. The highest  $|\Delta S_M^{\text{peak}}|$  value corresponds to the alloy with the largest  $M_{S,50}$  value (i.e.,  $M_{S,50} = 136 \text{ A}^2 \text{ kg}^{-1}$  for  $\text{Fe}_{88}\text{Zr}_7\text{B}_6\text{Cu}_1$ ). A similar  $|\Delta S_M^{\text{peak}}|$  vs.  $M_S$  relationship has been observed in other amorphous systems.<sup>18,19</sup> Owing to

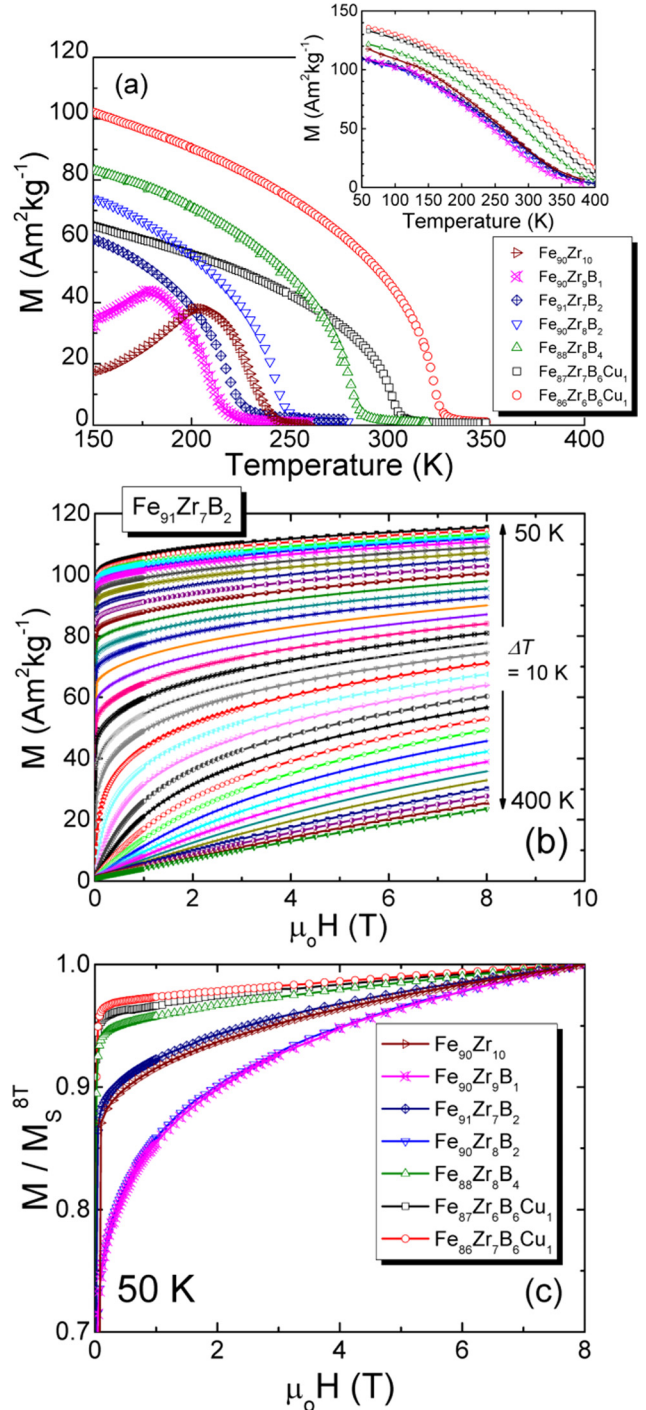
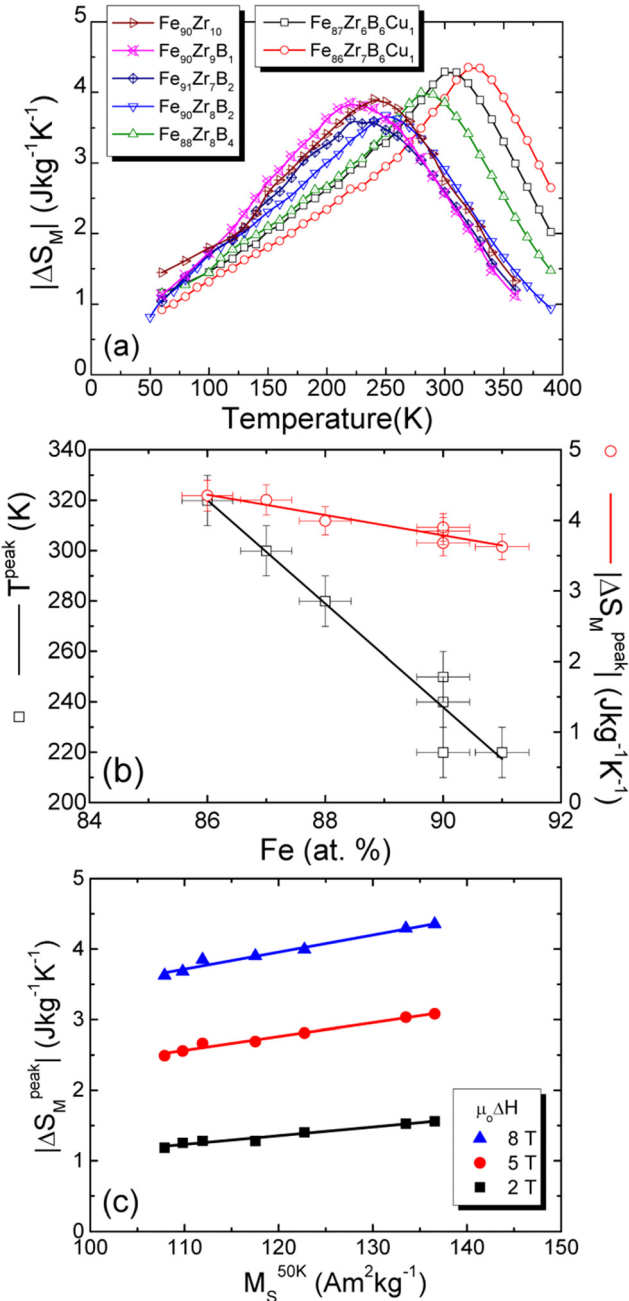
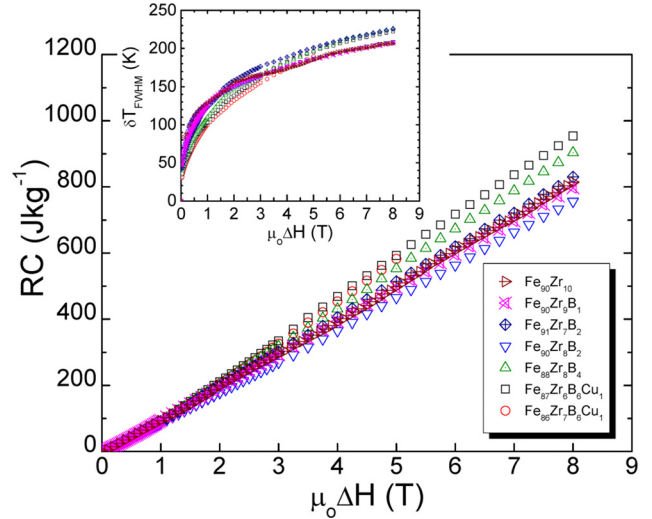


FIG. 1.  $M(T)$  curves measured under  $\mu_0H = 5$  mT (a), and  $M_S(T)$  curves obtained from the fitting of the  $M(\mu_0H)$  curves to the approach-to-saturation law (inset). (b) Set of  $M(\mu_0H)$  measured from 50 to 400 K for the  $\text{Fe}_{91}\text{Zr}_7\text{B}_2$  alloy. (c)  $M(\mu_0H)$  curves at 50 K normalized to the value at  $\mu_0H = 8$  T.

their inferior  $M_S$  values and a broader ferro-to-paramagnetic phase transition, these FeZrB(Cu) amorphous alloys exhibit a lower  $|\Delta S_M^{\text{peak}}|$  than that of pure Gd.<sup>20</sup> However, as shown in the inset of Fig. 3, the  $\delta T_{\text{FWHM}}$  values of these alloys for  $\mu_0\Delta H = 5$  T exceed 180 K [and continuously rises as  $\mu_0\Delta H$  does, because  $T_{\text{cold}}$  and  $T_{\text{hot}}$  go to lower and higher values, respectively], i.e., more than twice the working temperature span reported for Gd ( $\delta T_{\text{FWHM}} \sim 70$  K at 5 T).<sup>20</sup> These huge width of the  $|\Delta S_M(T)|$  curves explain why the  $RC$  values

TABLE I.  $|\Delta S_M^{peak}|$ ,  $\delta T_{FWHM}$ , and RC for  $\mu_0\Delta H = 2, 5$ , and 8 T for the studied amorphous alloys.

Sample	$T_C$ (K)	$ \Delta S_M^{peak} $ (J kg <sup>-1</sup> K <sup>-1</sup> )			$\delta T_{FWHM}$ (K)			RC (J kg <sup>-1</sup> )		
		2 T	5 T	8 T	2 T	5 T	8 T	2 T	5 T	8 T
Fe <sub>90</sub> Zr <sub>10</sub>	230 (5)	1.3	2.7	3.9	154	196	229	194	497	801
Fe <sub>90</sub> Zr <sub>9</sub> B <sub>1</sub>	210 (5)	1.3	2.7	3.8	153	190	215	198	492	795
Fe <sub>91</sub> Zr <sub>7</sub> B <sub>2</sub>	215 (5)	1.2	2.5	3.6	150	190	216	177	462	755
Fe <sub>90</sub> Zr <sub>8</sub> B <sub>2</sub>	240 (5)	1.3	2.6	3.7	158	201	225	198	514	830
Fe <sub>88</sub> Zr <sub>8</sub> B <sub>4</sub>	280 (5)	1.3	2.8	4.0	154	198	226	201	551	905
Fe <sub>87</sub> Zr <sub>6</sub> B <sub>6</sub> Cu <sub>1</sub>	300 (5)	1.6	3.0	4.3	143	197	226	208	590	953
Fe <sub>86</sub> Zr <sub>7</sub> B <sub>6</sub> Cu <sub>1</sub>	320 (5)	1.6	3.1	4.4	139	193	...	205	582	...

FIG. 2. (a)  $\Delta S_M(T)$  curves for a magnetic field change  $\mu_0\Delta H$  of 8 T. (b)  $|\Delta S_M^{peak}|$  and  $T^{peak}$  as a function of the Fe at. % for an applied magnetic field change  $\mu_0\Delta H = 8$  T. (c)  $|\Delta S_M^{peak}|$  vs.  $M_S$  at 50 K for  $\mu_0\Delta H = 2, 5$ , and 8 T. Lines are guides for the eyes.FIG. 3. Applied magnetic field change dependence of the refrigerant capacity, RC. Inset:  $\delta T_{FWHM}$  as a function of  $\mu_0\Delta H$ .

reach ca. 90% that for pure Gd (Ref. 20) [see Fig. 3 and Table I for the RC values under a magnetic field change of 2, 5, and 8 T].

The degree of asymmetry exhibited by the  $|\Delta S_M(T)|$  curves [see Fig. 2(a)] can be measured through the applied magnetic field dependence of  $(T_{hot} - T^{peak}) - (T^{peak} - T_{cold})$  [see inset in Fig. 4(b)]. In the case of positive values,  $|\Delta S_M|$  decreases slowly for  $T > T_C$ , whereas the negative values indicate a less marked variation of  $|\Delta S_M|$  in the magnetically ordered state ( $T < T_C$ ). The asymmetry is positive for all the studied samples under low magnetic field changes and reaches a maximum for a value of  $\mu_0\Delta H$  that depends on the alloy composition. However, only the alloys with 86 and 87 Fe at. % keep the positive values in the whole magnetic field range. As expected, the effect of these asymmetries also appears in the universal curve. Fig. 4(a) shows the rescaled  $\Delta S_M/\Delta S_M(\theta)$  curves for the Fe<sub>86</sub>Zr<sub>7</sub>B<sub>6</sub>Cu<sub>1</sub> amorphous alloy. In the magnetically ordered state, the  $\Delta S_M/\Delta S_M(\theta)$  curves for different values of the magnetic field change do not overlap for  $\theta < -1$ ; this behavior is present in all the studied ribbons, even for low values of the magnetic field change. These irregularities indicate the existence of magnetic anomalies at low temperatures,<sup>9</sup> due to either magnetic frustration or strong magneto-volume coupling. However, the curves match for  $\theta > -1$ , where magnetic frustration is no longer expected, thus validating the existence of a phenomenological universal curve. Moreover, all the rescaled  $\Delta S_M/\Delta S_M(\theta)$  curves for  $\mu_0\Delta H = 8$  T collapse in the normalized temperature range corresponding to the full width at half maximum ( $T_{cold}$  corresponds with  $\theta = -1$  and  $T_{hot}$  with  $\theta = 1$ ) and for higher temperatures, [see Fig. 4(b)], thus indicating a similar mechanism of the ferro-to-paramagnetic transition for the studied ribbons.

## SUMMARY AND CONCLUSIONS

We have studied the magnetic and MC properties of a set of FeZrB(Cu) amorphous alloys under applied magnetic field values up to  $\mu_0H = 8$  T. The saturation magnetization

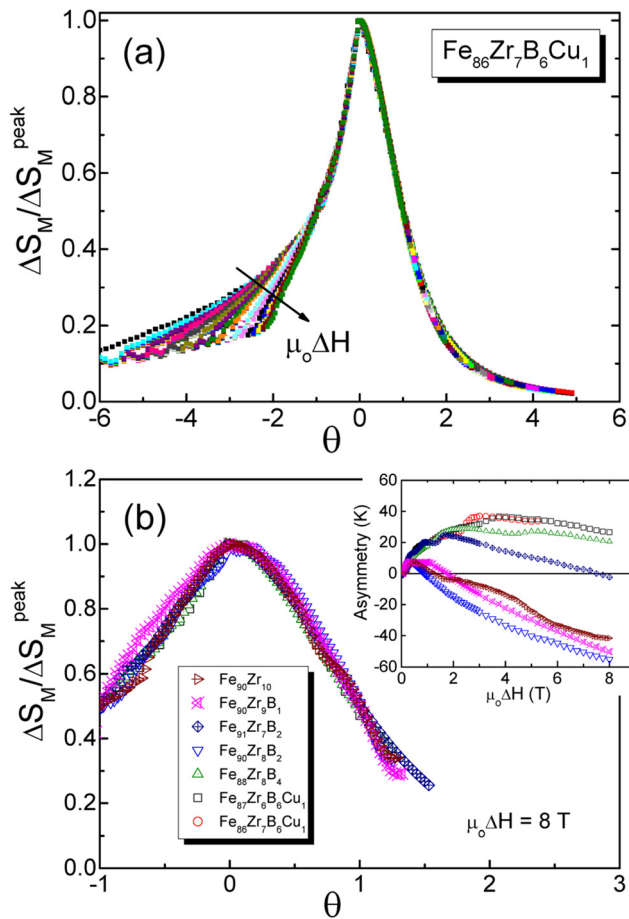


FIG. 4. (a) Rescaled  $\Delta S_M(\theta)$  curve of the  $\text{Fe}_{86}\text{Zr}_7\text{B}_6\text{Cu}_1$  ribbon (see text for details). (b) Comparison of the FeZrBCu alloys universal curves for  $\mu_0\Delta H = 8$  T in a rescaled temperature range. Inset: Magnetic field dependence of the asymmetry of the  $\Delta S_M(T)$  curves, defined as the difference  $(T_{\text{hot}} - T_{\text{peak}}) - (T_{\text{peak}} - T_{\text{cold}})$ .

and the temperature of the maximum of the magnetic entropy change decrease linearly with the increase of Fe-content. Although the maximum values for the magnetic entropy change are rather low (between  $3.6$  and  $4.4 \text{ J kg}^{-1} \text{ K}^{-1}$  for  $\mu_0\Delta H = 8$  T), the broad  $|\Delta S_M(T)|$  curves give rise to large  $RC(\mu_0\Delta H)$  values (ca. 90% of pure gadolinium at  $\mu_0\Delta H = 5$  T). We also tested the validity of the phenomenological universal curve under high-applied magnetic field values: when using two reference points, the curves overlap at rescaled temperatures  $\theta > -1$ , but the existence of magnetic frustration avoids the collapse in the low temperature

range. Nevertheless, the universal curves of all the samples overlap for  $\theta > -1$ , thus suggesting a similar magnetic behavior in the ferro-to-paramagnetic transition for these Fe-rich FeZr-based amorphous ribbons.

## ACKNOWLEDGMENTS

This work has been financially supported by: (a) projects CB-2010-01-156932 (CONACyT, Mexico), MAT2011-27573-C04 (MINECO, Spain), and IT711-13 (Basque Government, Spain); (b) Laboratorio Nacional de Investigaciones en Nanociencias y Nanotecnología (LINAN, IPICYT). C. F. Sánchez-Valdés thanks LINAN, IPICYT, and CONACyT (Project No. CB-2012-01-183770) for supporting his postdoctoral stay.

- <sup>1</sup>S. N. Kaul, V. Siruguri, and G. Chandra, *Phys. Rev. B* **45**, 12343 (1992).
- <sup>2</sup>J. M. Barandiarán, P. Gorria, I. Orue, M. L. Fdez-Gubieda, F. Plazaola, and A. Hernando, *Phys. Rev. B* **54**, 3026 (1996).
- <sup>3</sup>R. G. Calderón, L. F. Barquín, S. N. Kaul, J. C. Gómez Sal, P. Gorria, J. S. Pedersen, and R. K. Heenan, *Phys. Rev. B* **71**, 134413 (2005).
- <sup>4</sup>J. M. Barandiarán, P. Gorria, I. Orue, M. L. Fdez-Gubieda, F. Plazaola, J. C. Gómez Sal, L. F. Barquín, and L. Fournes, *J. Phys.: Condens. Matter* **9**, 5671 (1997).
- <sup>5</sup>P. Álvarez, P. Gorria, J. Sánchez Marcos, L. Fernández Barquín, and J. A. Blanco, *Intermetallics* **18**, 2464 (2010).
- <sup>6</sup>A. M. Tishin and Y. I. Spichkin, *The Magnetocaloric Effect and its Applications* (IOP Publishing, Bristol, 2003).
- <sup>7</sup>P. Álvarez, J. L. Sánchez Llamazares, P. Gorria, and J. A. Blanco, *Appl. Phys. Lett.* **99**, 232501 (2011).
- <sup>8</sup>P. Alvarez, P. Gorria, J. L. Sánchez Llamazares, and J. A. Blanco, *J. Alloys Compd.* **568**, 98 (2013).
- <sup>9</sup>K. A. Gschneidner, Jr., V. K. Pecharsky, and A. O. Tsokol, *Rep. Prog. Phys.* **68**, 1479 (2005).
- <sup>10</sup>P. Gorria, J. L. Sánchez Llamazares, P. Álvarez, M. J. Pérez, J. Sánchez Marcos, and J. A. Blanco, *J. Phys. D: Appl. Phys.* **41**, 192003 (2008).
- <sup>11</sup>P. Álvarez, J. Sánchez-Marcos, P. Gorria, L. Fernández Barquín, and J. A. Blanco, *J. Alloys Compd.* **504**, S150 (2010).
- <sup>12</sup>V. Franco, J. S. Blázquez, M. Millán, J. M. Borrego, C. F. Conde, and A. Conde, *J. Appl. Phys.* **101**, 09C503 (2007).
- <sup>13</sup>V. Franco, A. Conde, V. Provenzano, and R. D. Shull, *J. Magn. Magn. Mater.* **322**, 218–223 (2010).
- <sup>14</sup>V. Franco, J. M. Borrego, A. Conde, and S. Roth, *Appl. Phys. Lett.* **88**, 132509 (2006).
- <sup>15</sup>V. Franco and A. Conde, *Int. J. Refrig.* **33**, 465 (2010).
- <sup>16</sup>V. Franco, R. Caballero-Flores, A. Conde, Q. Y. Dong, and H. W. Zhang, *J. Magn. Magn. Mater.* **321**, 1115 (2009).
- <sup>17</sup>B. D. Cullity and C. D. Graham, *Introduction to Magnetic Materials* (IEEE Press-Wiley, 2009).
- <sup>18</sup>J. Y. Law, R. V. Ramanujan, and V. Franco, *J. Alloys Compd.* **508**, 14 (2010).
- <sup>19</sup>V. Franco, C. F. Conde, J. S. Blázquez, A. Conde, P. Švec, D. Janičkovič, and L. F. Kiss, *J. Appl. Phys.* **101**, 093903 (2007).
- <sup>20</sup>J. F. Elliott, S. Legvold, and F. H. Spedding, *Phys. Rev.* **91**, 28 (1953).